

Quantum Dot Solar Cells

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The University of Toledo

April 11, 2011

Condensed Matter Seminar, Department of Physics

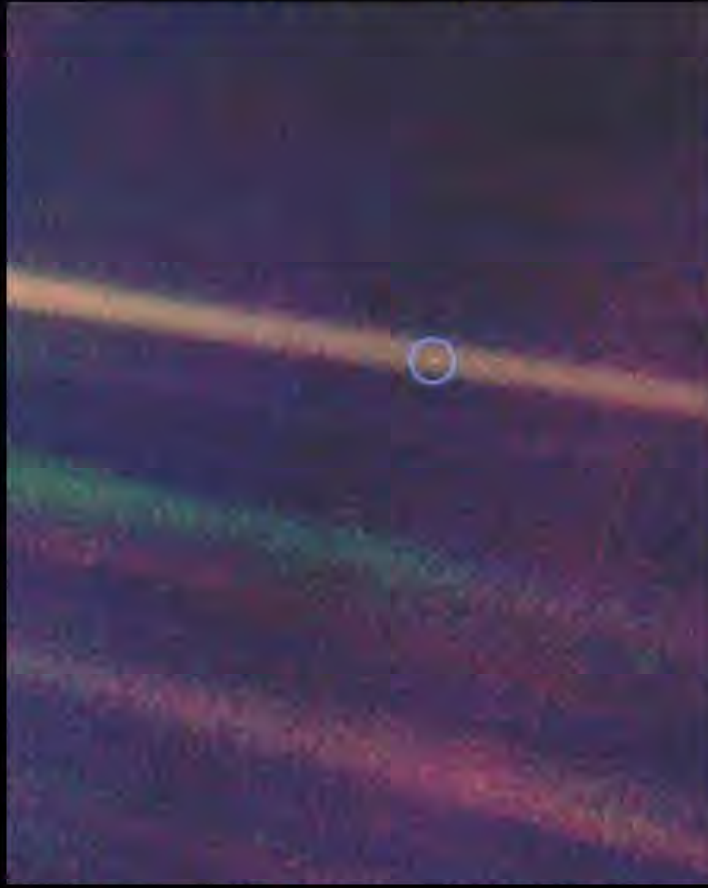
Case Western Reserve University



Our little dot



In 1990, Carl Sagan convinced NASA engineers to turn Voyager for one last, homeward look before leaving the solar system



“Look again at that dot. That’s here. That’s home. That’s us. On it, everyone you love, everyone you know, everyone you ever heard of, every human being who ever was, lived out their lives Every hunter and forager, every hero and coward, every creator and destroyer of civilization, every king and peasant, every young couple in love, every mother and father, hopeful child, inventor and explorer, every teacher of morals, ... every saint and sinner in the history of our species lived there--on a mote of dust suspended in a sunbeam.”

“....The Earth is the only world known so far to harbor life. There is nowhere else, at least in the near future, to which our species could migrate ... Like it or not, for the moment the Earth is where we make our stand.”

-- Energy Secretary, Steve Chu

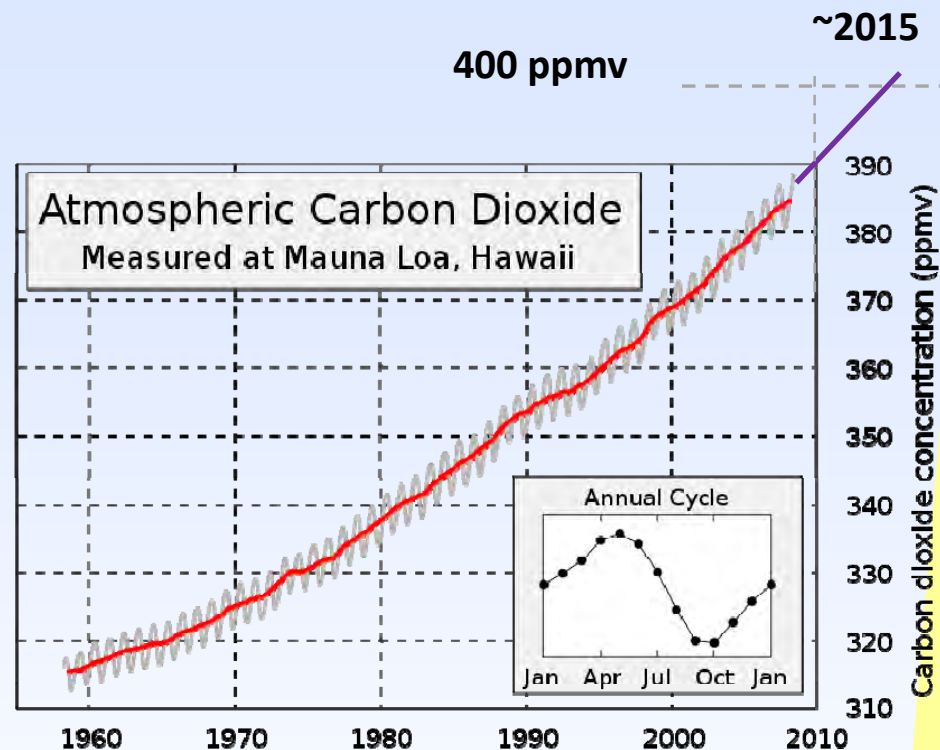
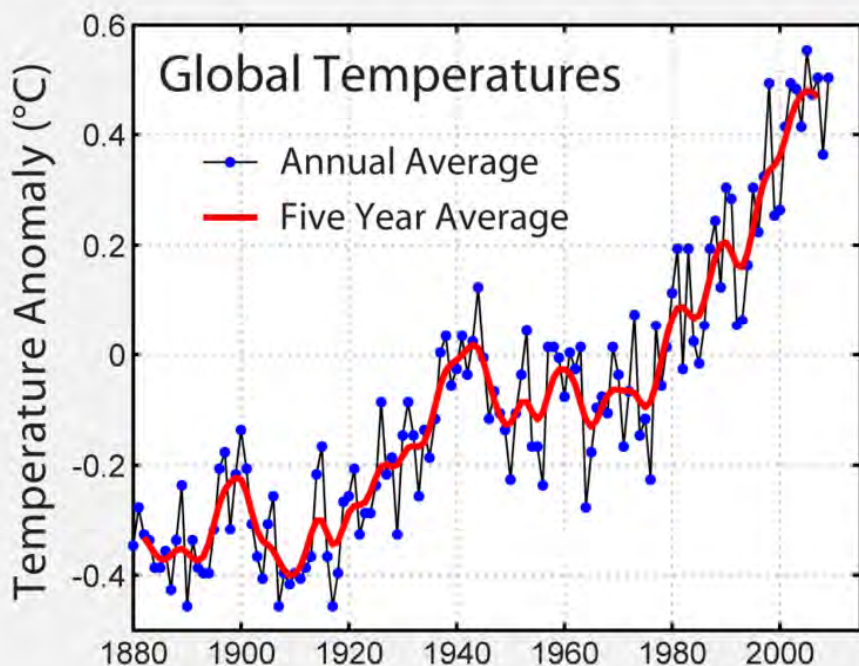
\$ per Watt Workshop held Aug. 11-12, 2010 [http://www1.eere.energy.gov/solar/dollar_per_watt.html].



On watch: global temperatures, atmospheric CO₂



Global average temperatures from NASA's *Goddard Institute for Space Studies* (Columbia University). Data set follows methodology developed by J. Hansen.



Keeling curve, data from Mauna Loa, Hawaii.

Hansen, J., et al. (2006) "Global temperature change", PNAS 103: 14288-14293.





Need for clean energy



NATURE | VOL 395 | 29 OCTOBER 1998

Energy implications of future stabilization of Atmospheric CO₂ content

M. Hoffert et al.

Growth

- Global energy consumption -- 1.6-1.7% per year.
- Includes for 1%/yr. efficiency improvement
- 28 TW global power consumption by 2050
- Population growth primarily in less-developed countries → increased C-intensity.
- **~15 TW of the ~30 TW should come from C-free sources in 2050.**

Health

Coal-fired power plants:

- 59% of total U.S. sulfur dioxide pollution
- 18% of total nitrous oxides every year
- largest polluter of toxic mercury pollution

All U.S. power plants: release over 40% of U.S. CO₂

[Sources – U.S. DOE and U.S. EPA]

Acid rain, smog (ozone), soot →
unhealthy ecosystems, respiratory
problems, unhealthy lungs (incl.
asthma)

A developmental toxin,
affecting unborn children





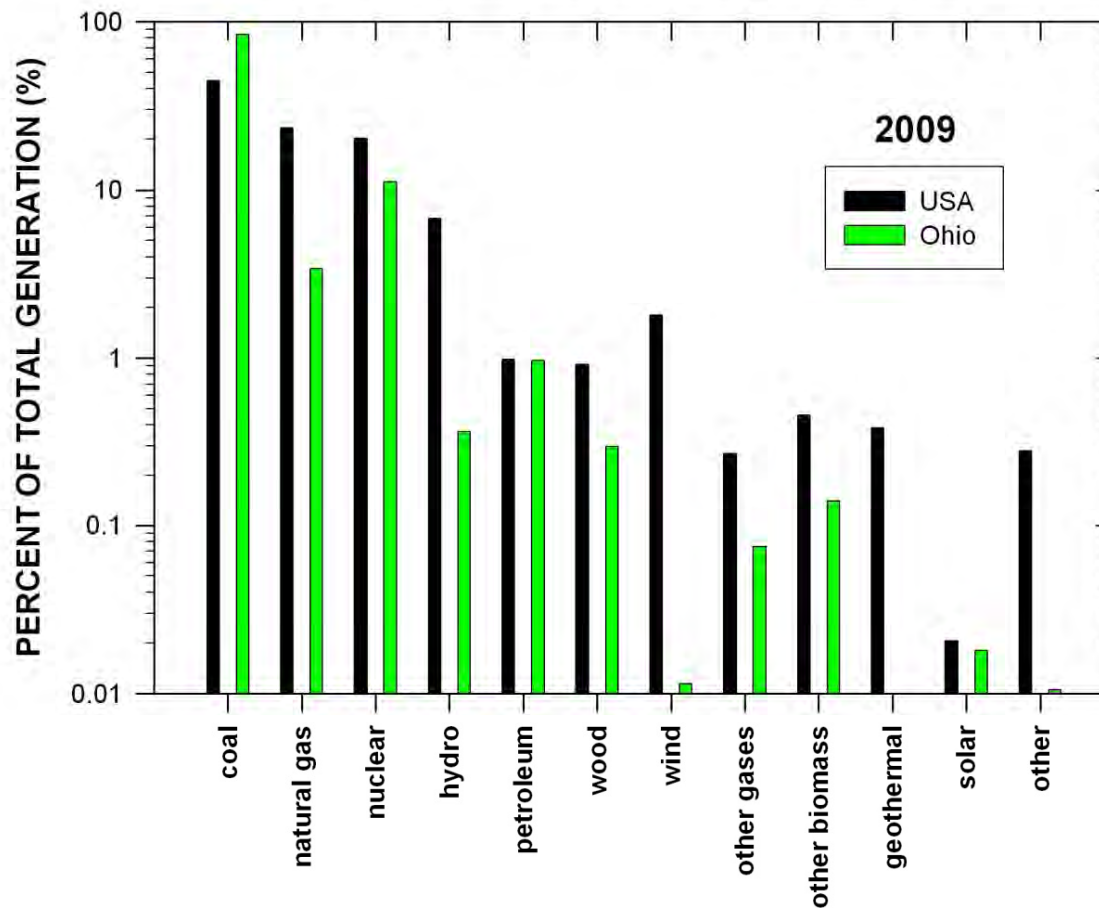
Volume of Earth: $1.1 \times 10^{12} \text{ km}^3$
Volume of water: $1.4 \times 10^9 \text{ km}^3$
Volume of atmosphere: $4.2 \times 10^9 \text{ km}^3$



How does Ohio's electricity generation stack up?



CONTRIBUTIONS OF FUEL SOURCES TO TOTAL ELECTRIC GENERATION



Data Source: Energy Information Administration
U.S. Department of Energy

Graphs prepared by B. Martner,
Lafayette, CO



Happy Birthday (April 11) to Uncle Brooks!



Brooks Martner (retired) and
Marcia Politovich (not retired)
with their ~5 kW PV (c-Si) system





Dirty energy, toxic food, air, and water

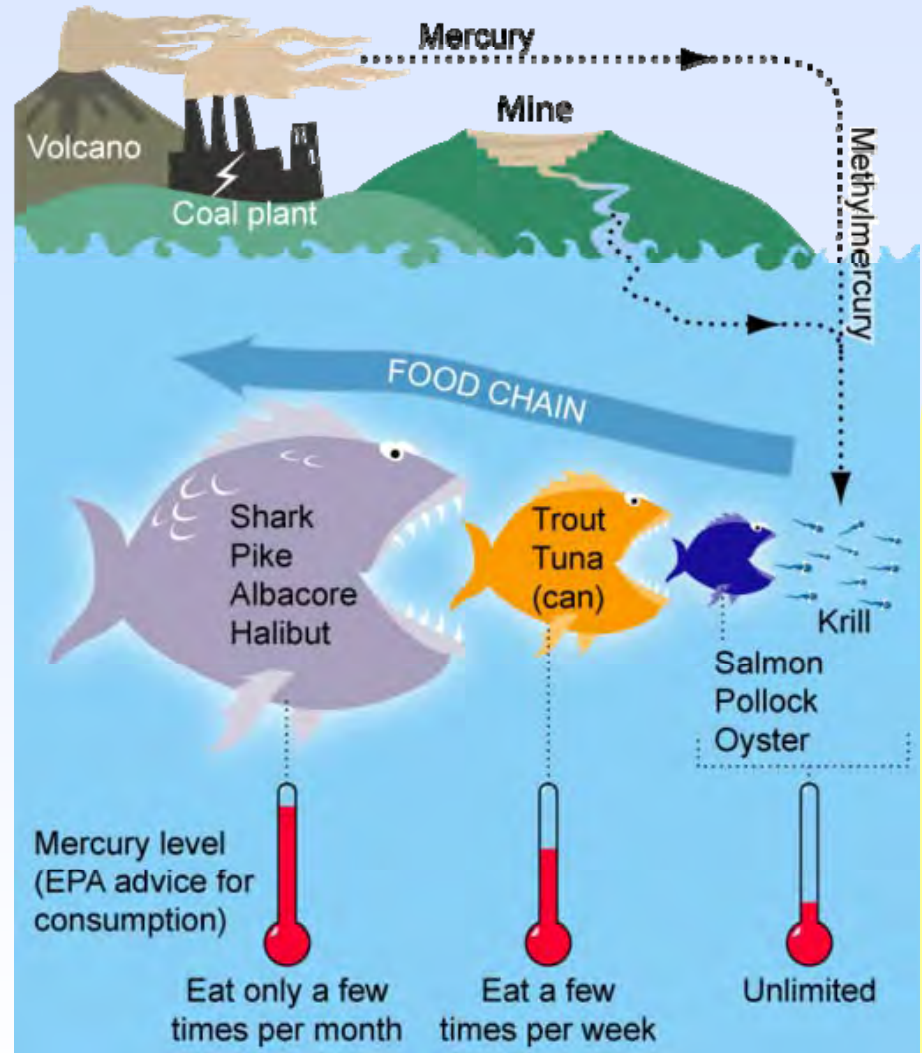


Burn a 100 W light bulb for a year:

$$\frac{876 \text{ kW} \cdot \text{h}}{2.0 \text{ kW} \cdot \text{h/kg}} = 438 \text{ kg of coal} = 966 \text{ pounds of coal}$$

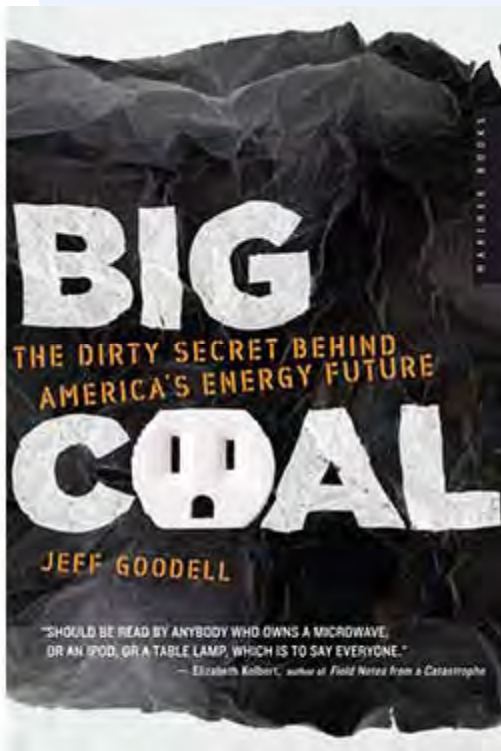
Ohio only ~85% coal, but also 5-10% transmission losses → ~900 lbs. (~400 kg) coal to light that bulb (or four 25 W CFLs) in Ohio.

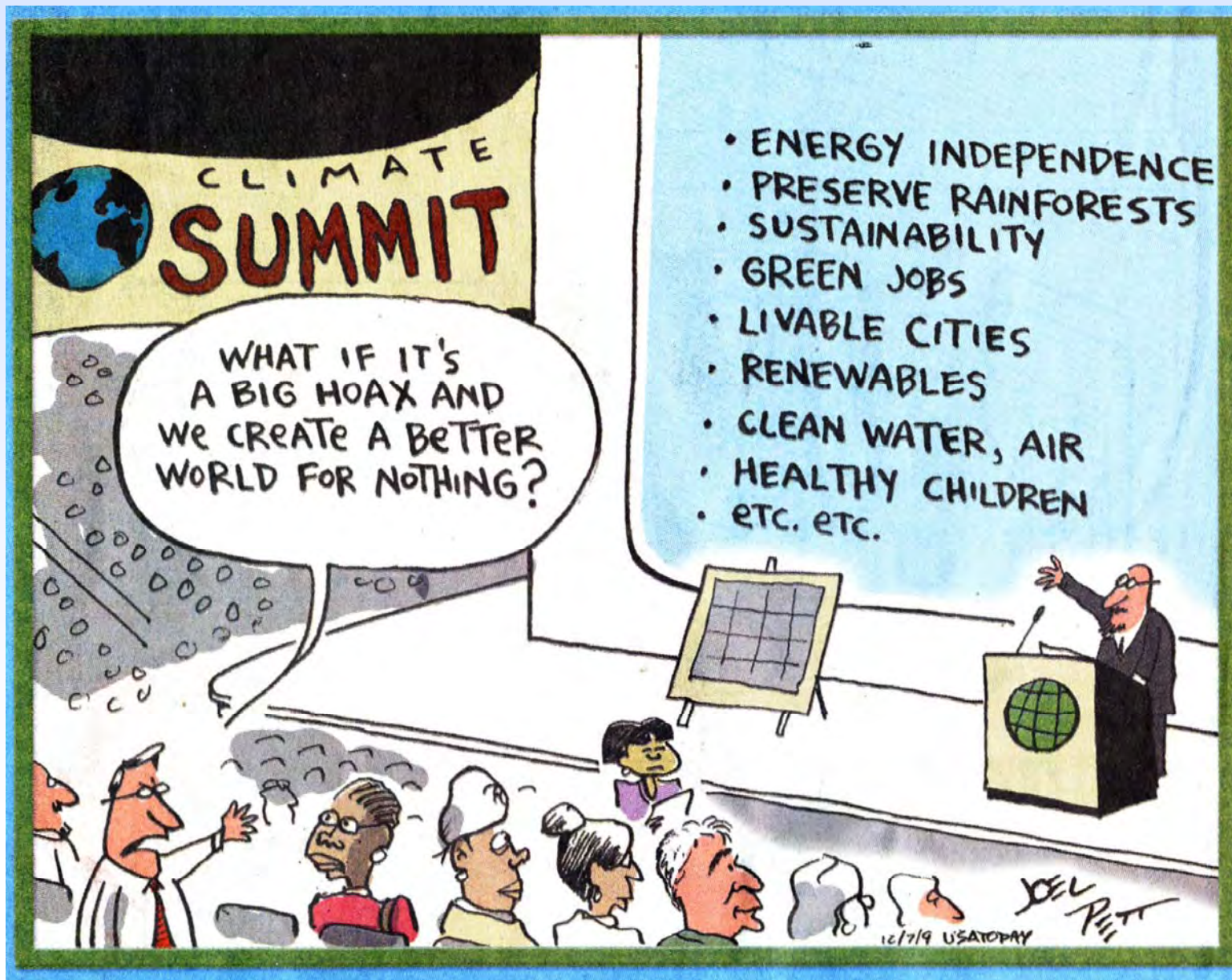
<http://en.wikipedia.org/wiki/Coal>



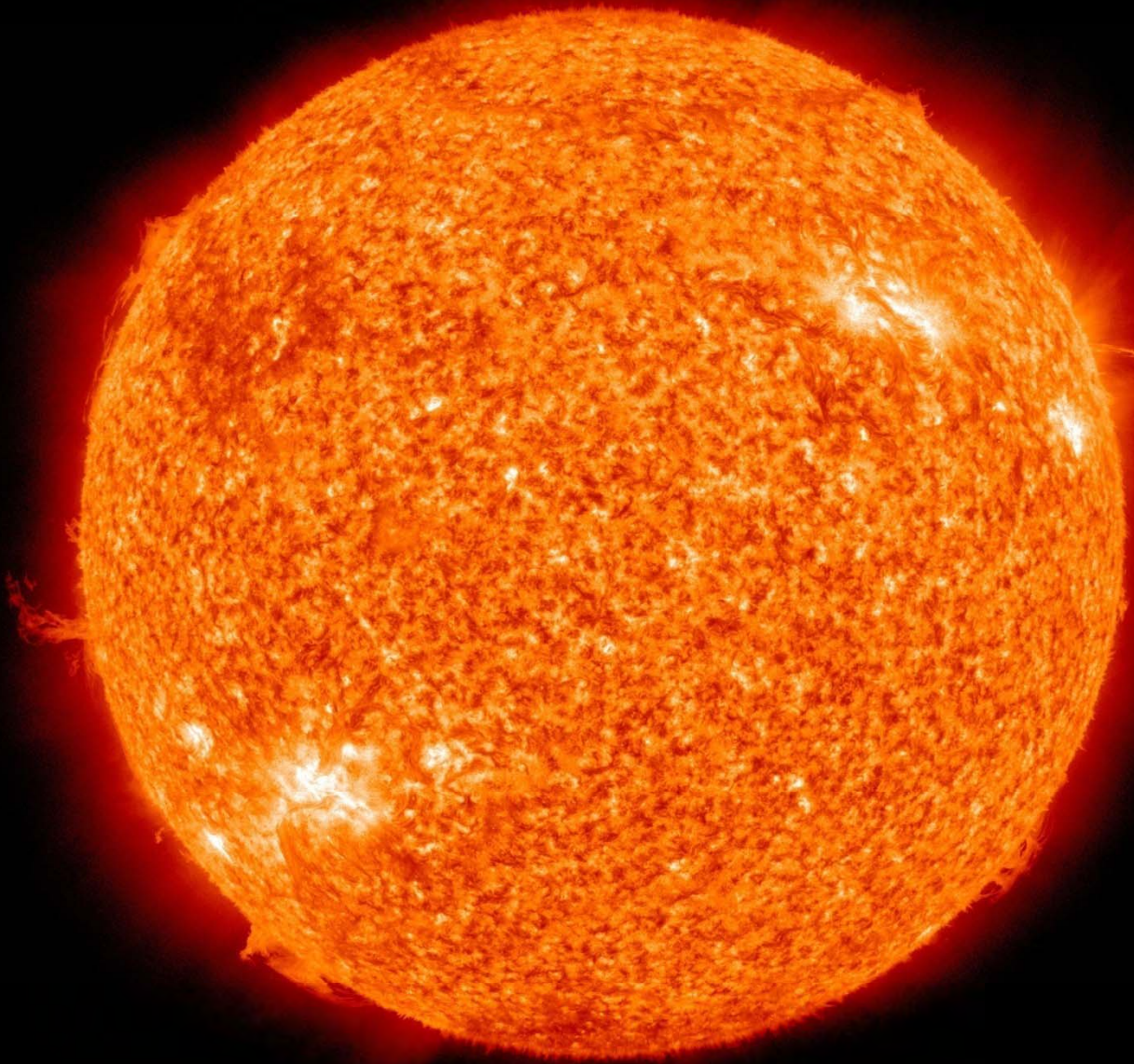
Coal energy density:
~24 MJ/kg

http://en.wikipedia.org/wiki/Mercury_in_fish





The Sun (worth revering)



“Why Does the Sun Shine?”
by *They Might Be Giants*

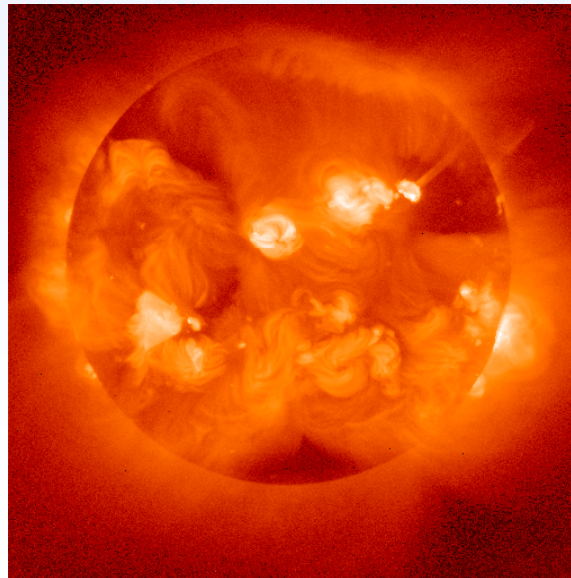
The sun is a mass of
incandescent gas
A gigantic nuclear furnace
Where hydrogen is built into
helium
At a temperature of millions
of degrees

Yo ho, it's hot, the sun is not
A place where we could live
But here on Earth there'd be
no life
Without the light it gives

We need its light
We need its heat
We need its energy
Without the sun, without a
doubt
There'd be no you and me

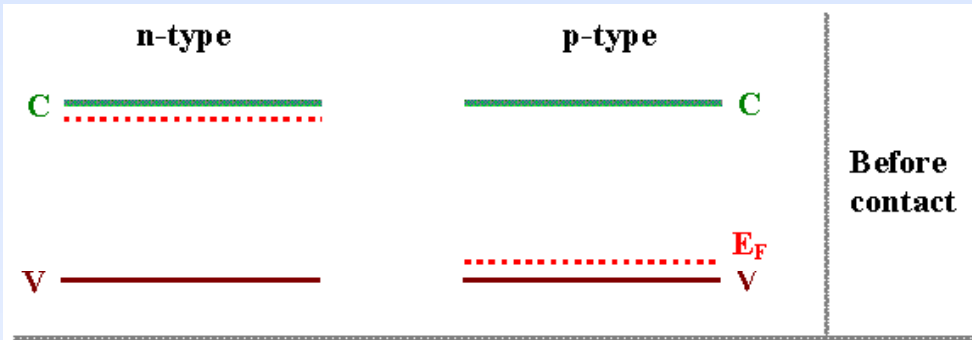


- Theoretical: 1.2×10^5 TW solar energy potential
(1.76×10^5 TW striking Earth; 0.30 Global mean albedo)
- Energy in 1 hr of sunlight \leftrightarrow 14 TW for a year
- Practical: > On-shore electricity generation potential of \approx 600 TW
(10% conversion efficiency).
- *Photosynthesis*: 170 TW (90 TW land-based, and 80 TW aquatic)

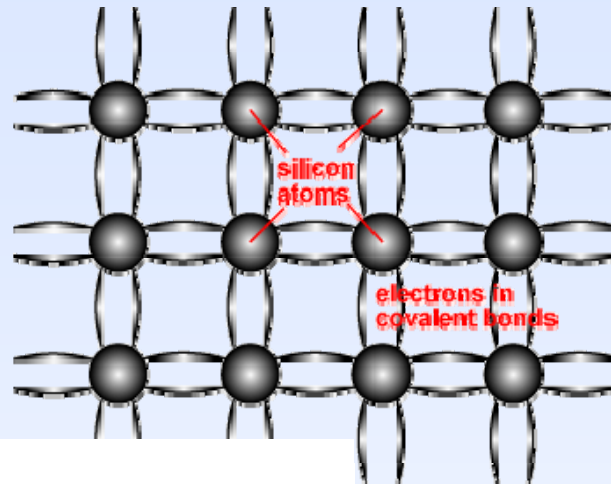




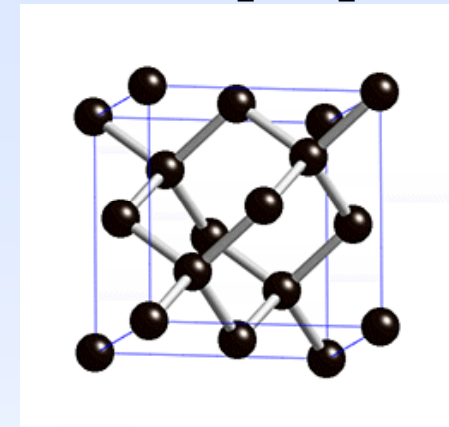
Sunlight to electricity (photovoltaic conversion)



Doping a semiconductor (e.g., Si)

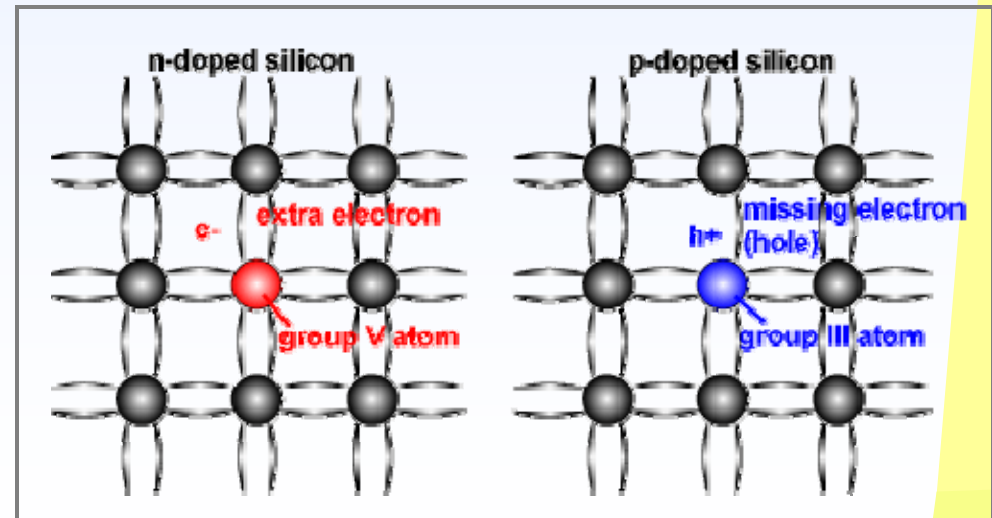


en.wikipedia.org/wiki/File:Diamond_Cu
bic-F_lattice_animation.gif



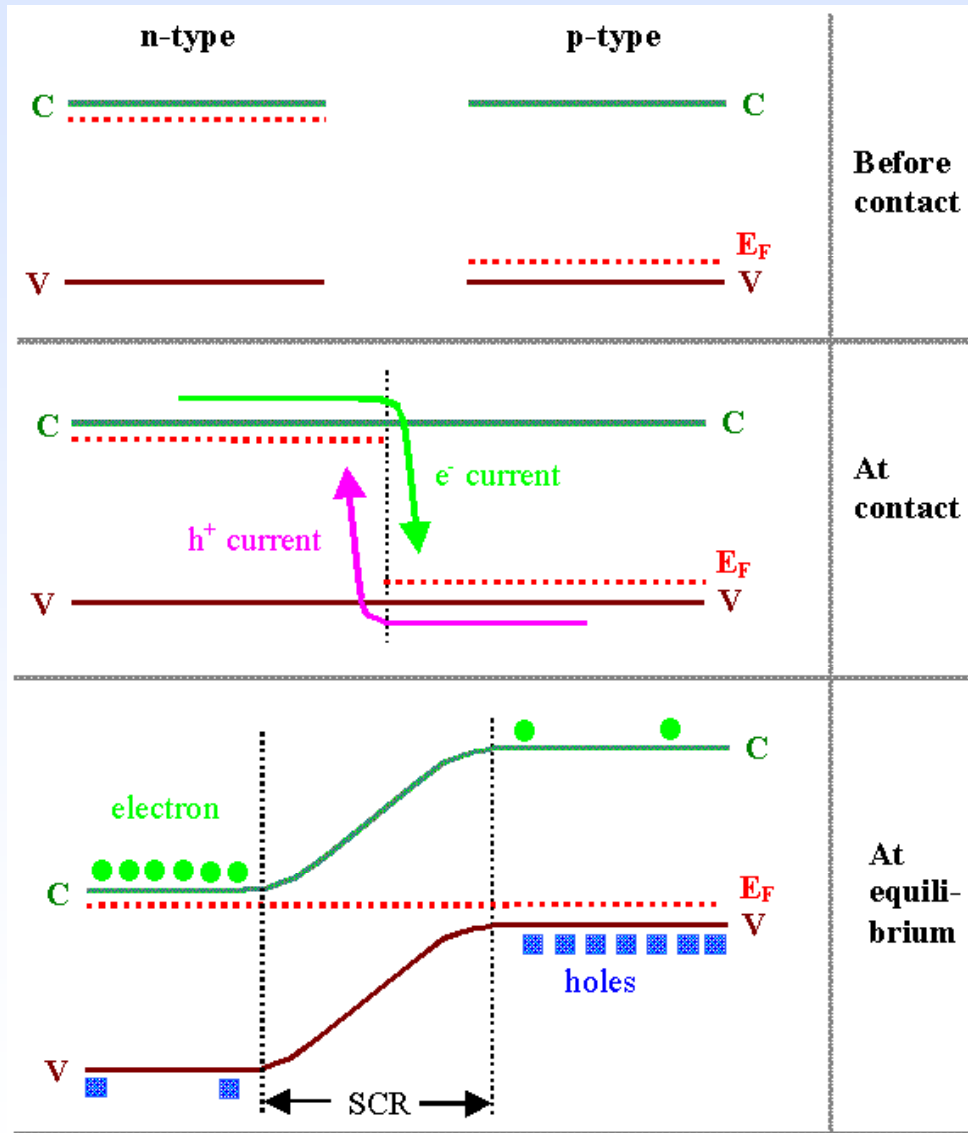
Diamond cubic crystal structure

<p>3A</p> <p>5 B</p> <p>[He]2s²2p¹ boron 10.81</p>	<p>4A</p> <p>6 C</p> <p>[He]2s²2p² carbon 12.01</p>	<p>5A</p> <p>7 N</p> <p>[He]2s²2p³ nitrogen 14.01</p>
<p>13 Al</p> <p>[Ne]3s²3p¹ aluminum 26.98</p>	<p>14 Si</p> <p>[Ne]3s²3p² silicon 28.09</p>	<p>15 P</p> <p>[Ne]3s²3p³ phosphorus 30.97</p>

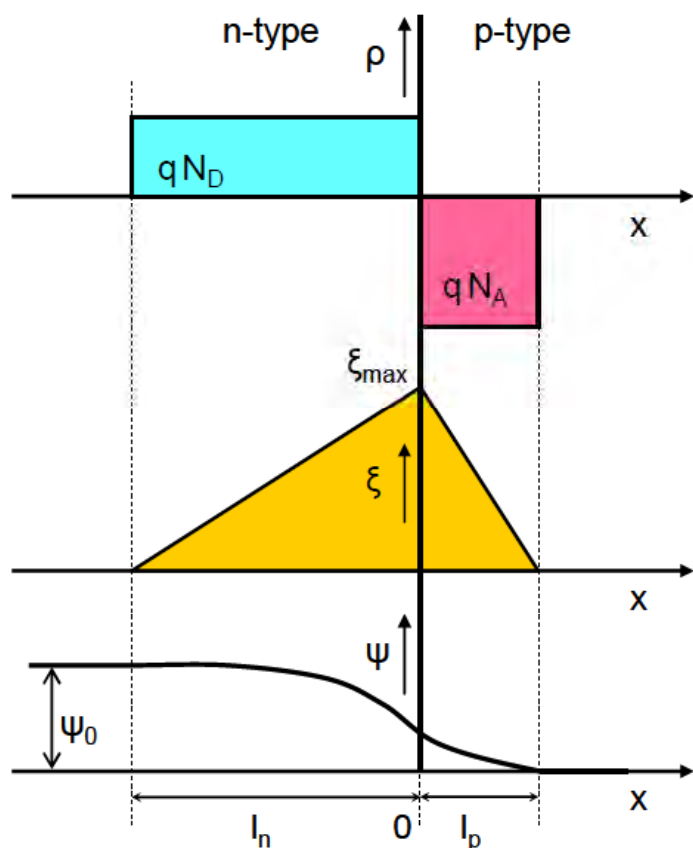




Sunlight to electricity (photovoltaic conversion)



http://www.tf.uni-kiel.de/matwis/amat/semi_en/kap_2/backbone/r2_2_4.html



Transport of free carriers occurs via drift (electric field) and diffusion (concentration gradient)

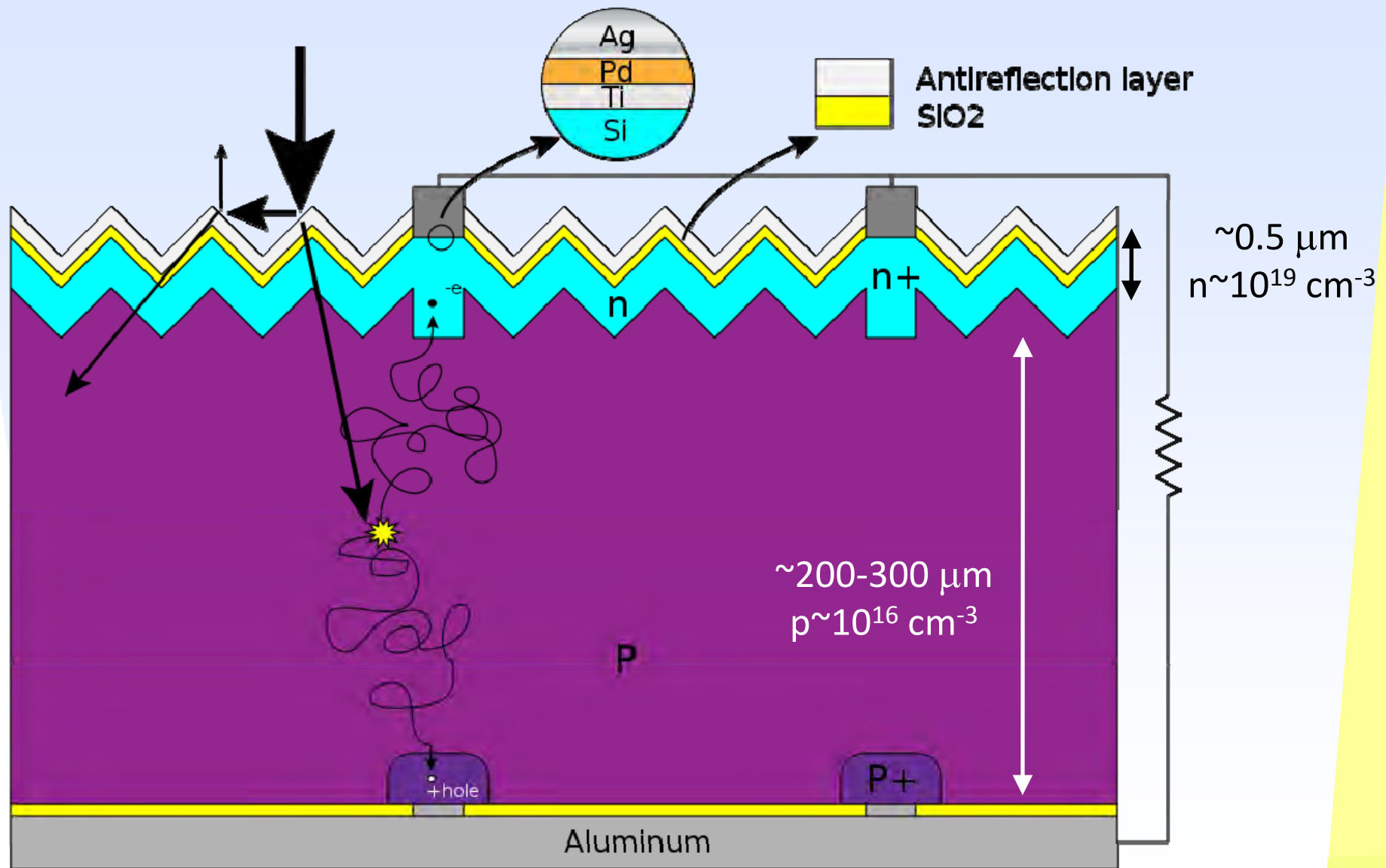
$$J_n = q\mu_n n(x)E + qD_n \frac{dn(x)}{dx}$$

$$J_p = q\mu_p p(x)E + qD_p \frac{dp(x)}{dx}$$

Top: space-charge density (ionized dopants)

Middle: electric field

Bottom: electrostatic potential





Estimating depletion width for c-Si



$$W = \left[\frac{2K_s \epsilon_0}{q} \left(\frac{N_A + N_D}{N_A N_D} \right) (V_{bi} - V) \right]^{1/2}$$

$$K_s \approx 11.9$$

$$\epsilon_0 = 8.85 \times 10^{-12} \text{ C V}^{-1} \text{ m}^{-1}$$

$$\frac{10^{16} \text{ cm}^{-3} + 10^{19} \text{ cm}^{-3}}{10^{35} \text{ cm}^{-6}} \approx 10^{-16} \text{ cm}^3$$

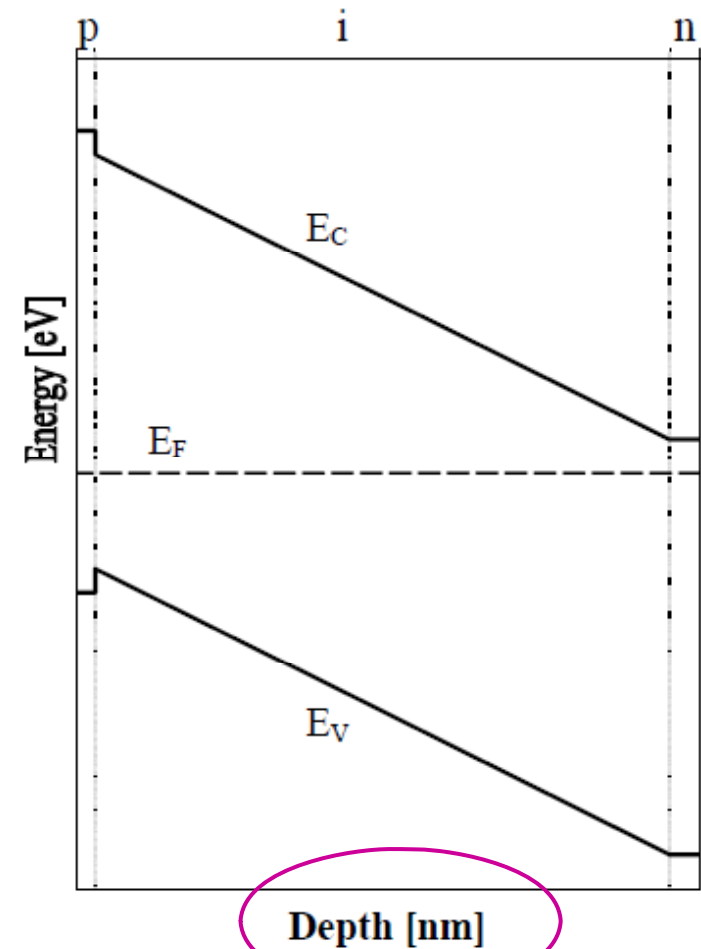
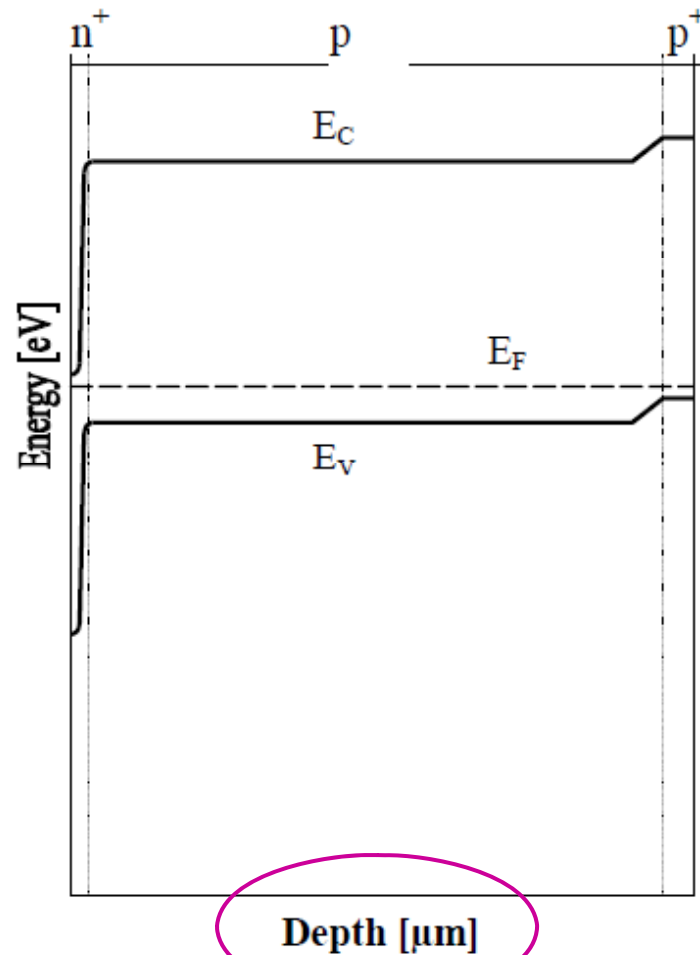
$$W \approx \left[1.3 \times 10^{-13} \text{ V}^{-1} \text{ m}^2 (V_{bi} - V) \right]^{1/2}$$

$$V_{bi} = \frac{kT}{q} \ln \left(\frac{N_A N_D}{n_i^2} \right)$$

$$n_i(\text{c-Si}) \approx 1.5 \times 10^{10} \text{ cm}^{-3}$$

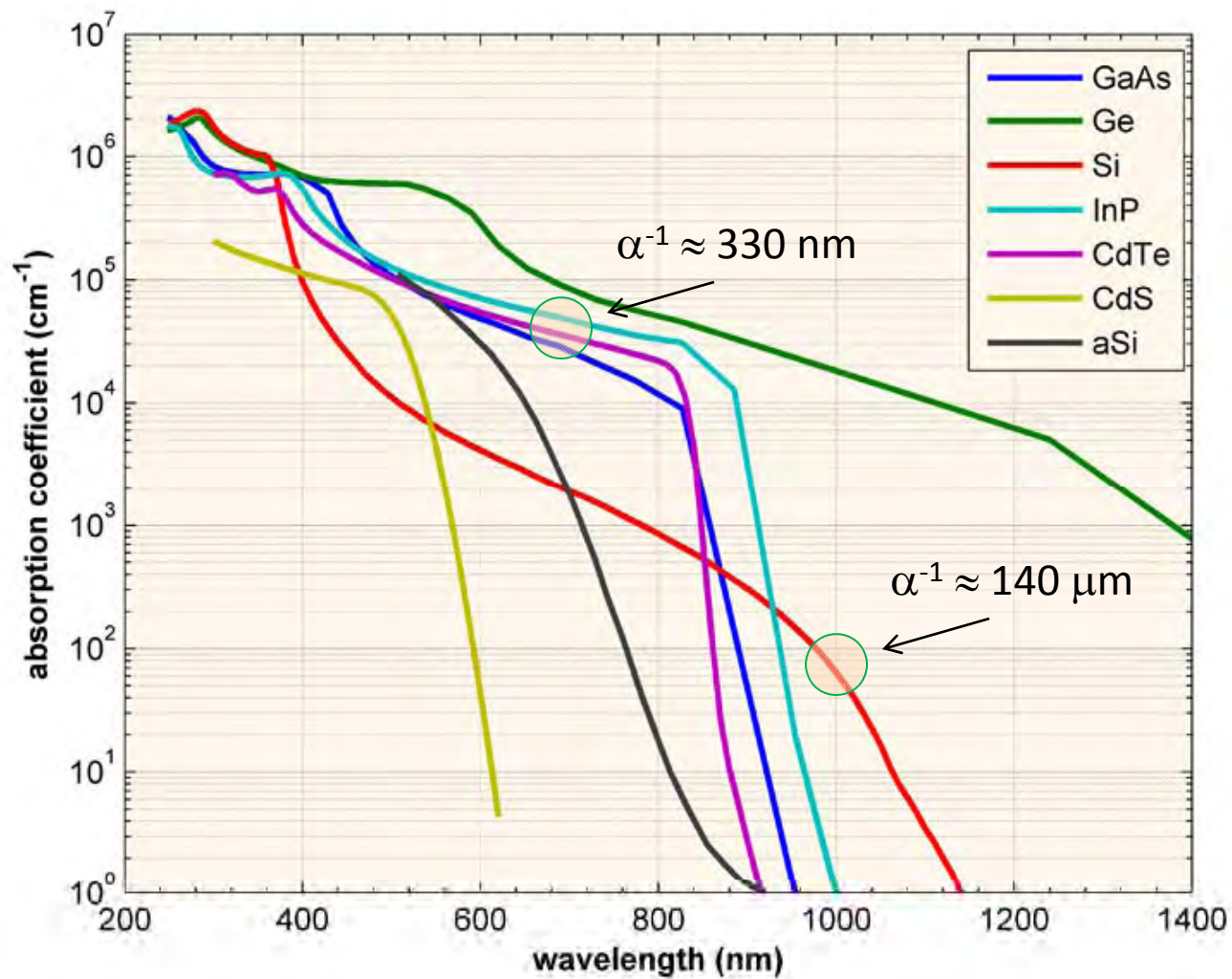
$$V_{bi} \approx \frac{kT}{q} \ln \left(\frac{10^{16} \text{ cm}^{-3} 10^{19} \text{ cm}^{-3}}{2.2 \times 10^{20} \text{ cm}^{-3}} \right) \approx 0.84 \text{ V}$$

$$W \approx 330 \text{ nm}$$



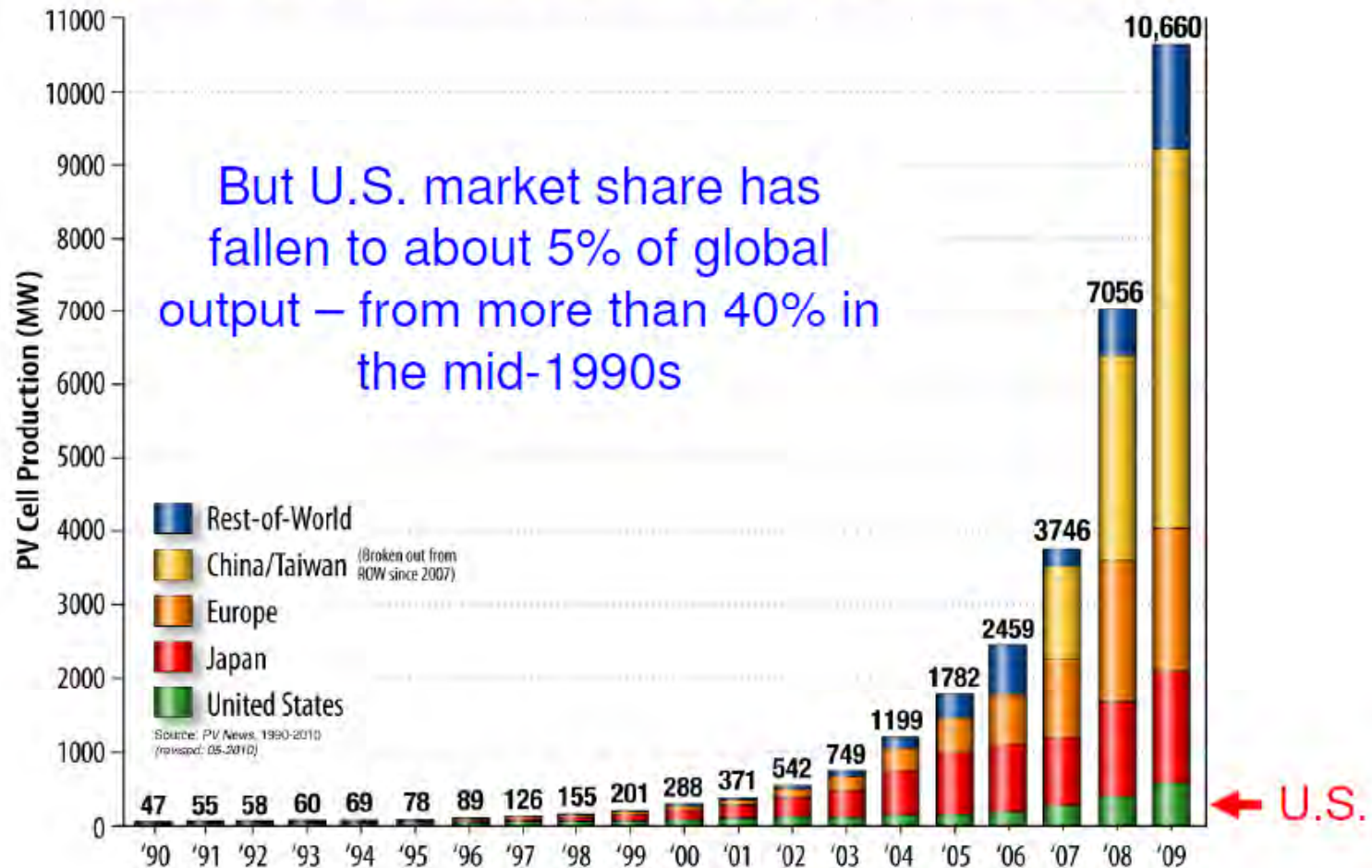
(a) c-Si solar cell (indirect gap), and (b) a-Si solar cell (direct gap)

$$I(\lambda, x) = I_0 e^{-\alpha(\lambda)x}$$

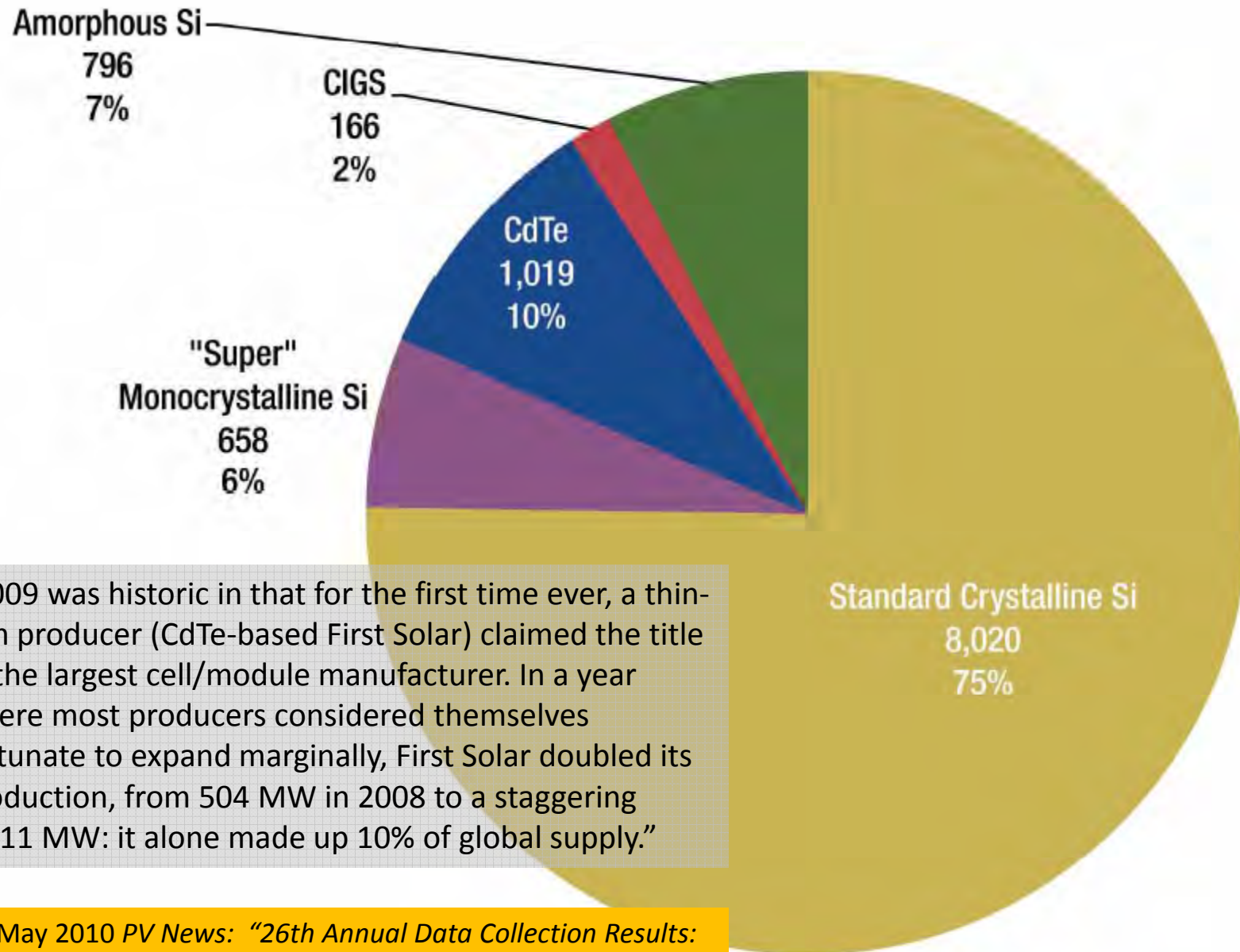




Solar PV is a booming global industry



Worldwide production of solar photovoltaics – in Megawatts



“2009 was historic in that for the first time ever, a thin-film producer (CdTe-based First Solar) claimed the title of the largest cell/module manufacturer. In a year where most producers considered themselves fortunate to expand marginally, First Solar doubled its production, from 504 MW in 2008 to a staggering 1,011 MW: it alone made up 10% of global supply.”

from May 2010 *PV News*: “26th Annual Data Collection Results: Another Bumper Year for Manufacturing Masks Turmoil”

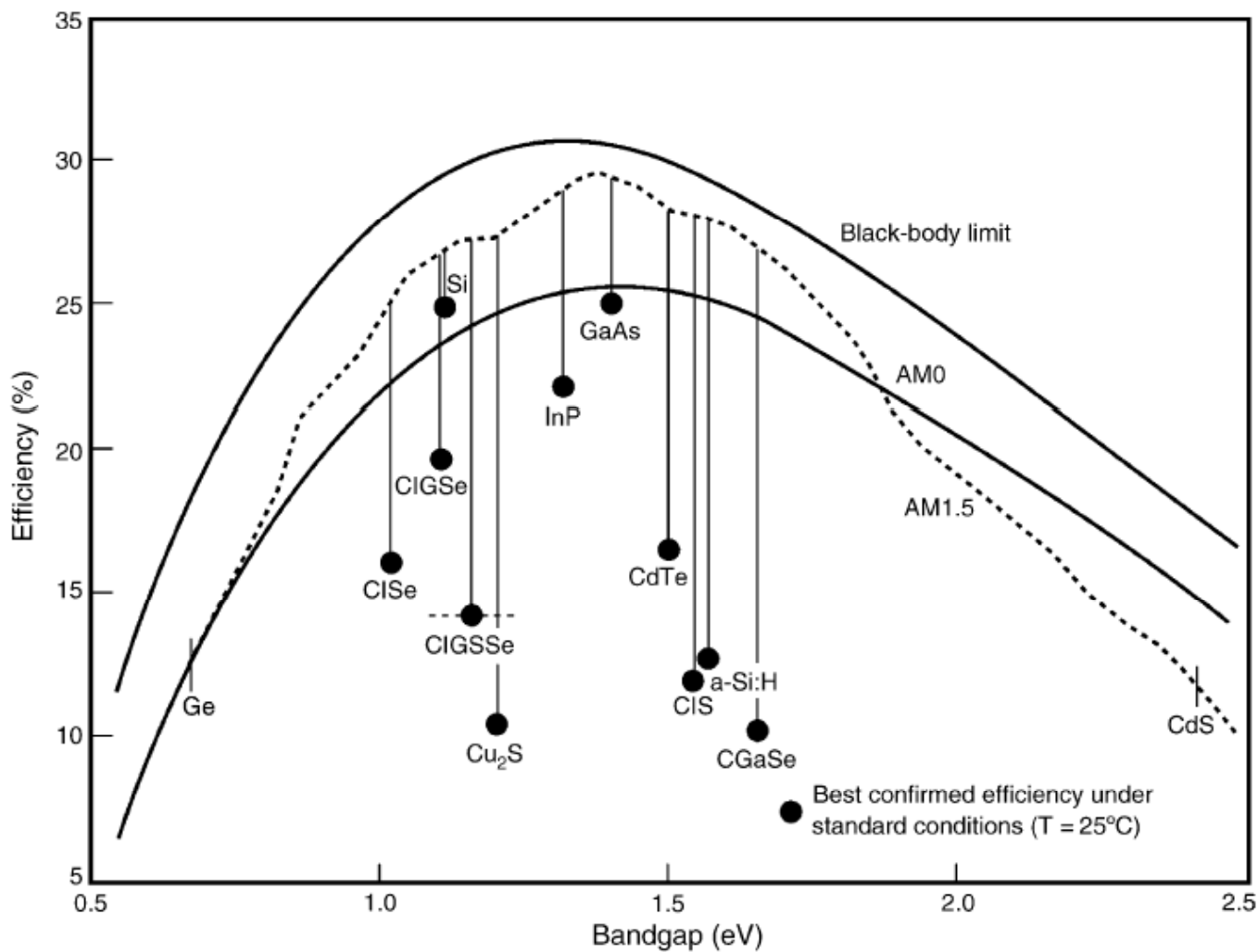


Fig. 3. Performance gaps between best device efficiencies in the laboratory and attainable efficiencies for several solar cell technologies.



Attained vs. attainable open circuit photovoltage



Cell Type	E_g at RT (eV)	V_{OC}^{MAX} (V)	V_{OC} (V)	V_{OC} loss (V)	V_{OC}/V_{OC}^{MAX} (%)
SC-Si	1.12	0.84	0.71	0.13	85
GaAs	1.42	1.14	1.02	0.12	90
InP	1.28	1.00	0.88	0.12	88
CdTe	1.45	1.17	0.84	0.33	72
CIGS	1.14	0.86	0.72	0.14	84
a-Si	1.7	1.42	0.86	0.56	61
DSS					
(black dye)	1.4	1.12	0.72	0.40	64
(Red N719)	1.6	1.32	0.85	0.47	64
(Red N3)	2.0	1.72	0.80	0.92	47
OPV	1.55	1.27	0.75	0.52	59



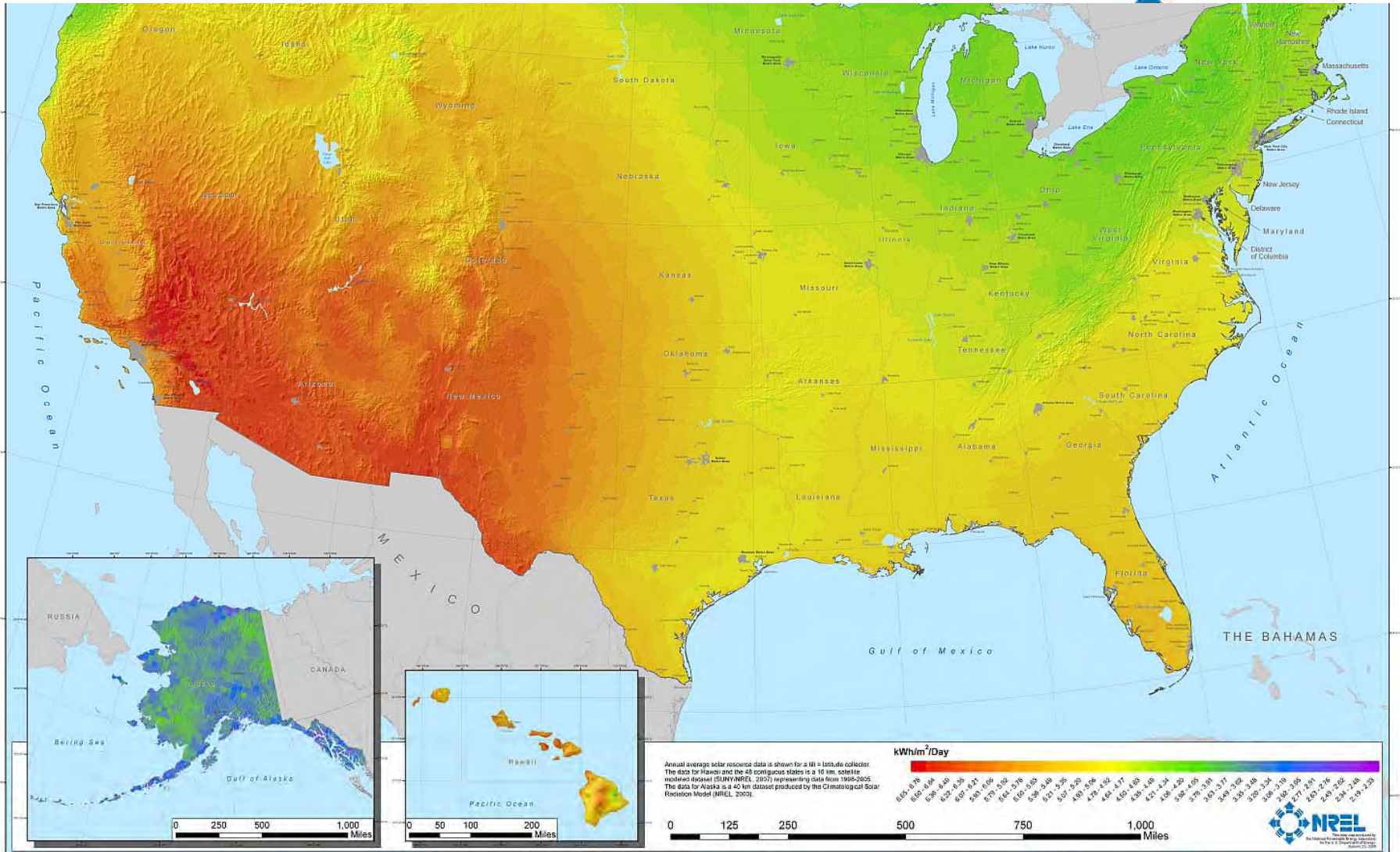
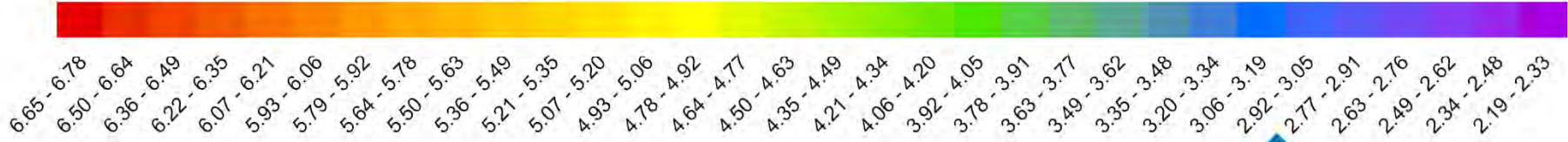
Attained vs. attainable short-circuit photocurrent



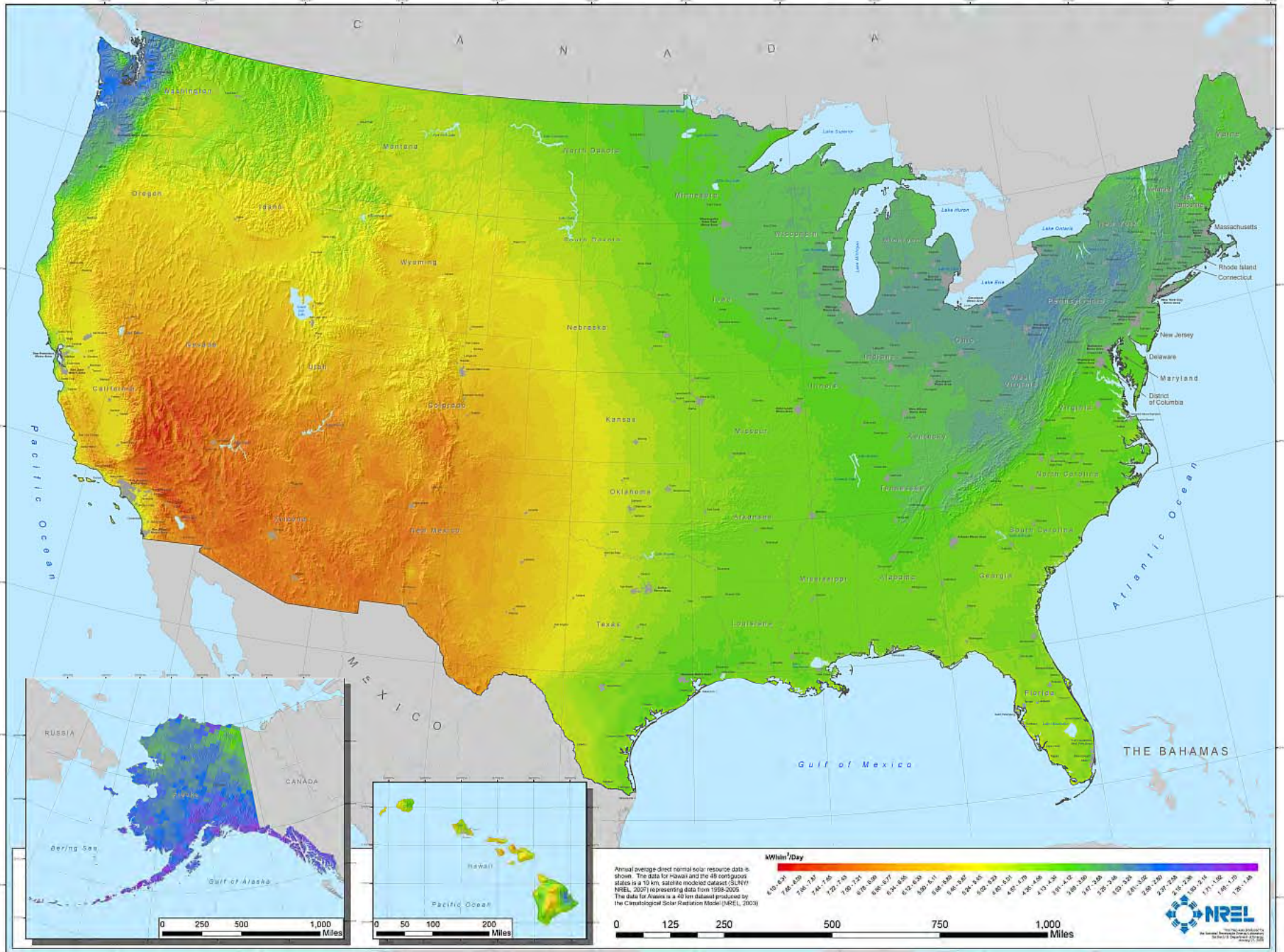
Cell Type	E_g at RT (eV)	J_{sc}^{MAX} (mA/cm ²)	J_{sc} (mA/cm ²)	J_{sc}/J_{sc}^{MAX} (%)
SC-Si	1.12	43.8	42.7	98
GaAs	1.42	32.0	28.5	89
InP	1.28	36.3	29.5	81
CdTe	1.45	30.8	25.9	84
CIGS	1.15	42	33.5	80
a-Si	1.7	22.4	17.5	78
DSS (black dye)	1.4	33.3	20.5	62
(Red N719)	1.6	25.5	17.7	70
(Red N3)	2.0	14.4	9.2	64
OPV	1.55	26.9	14.7	55

United States Photovoltaic Solar Resource : Flat Plate Tilted at Latitude

kWh/m²/Day



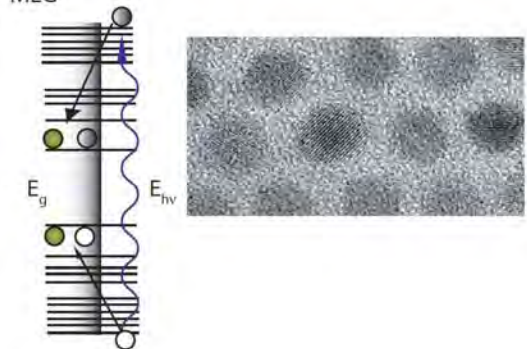
United States Concentrating Solar Power Resource : Direct Normal





- High surface area devices →
 - strong light absorption (dye-sensitized nanostructured TiO₂)
 - fast charge separation (proximity of photoexcited carriers to charge-separating interface)
- Customizable properties →
 - Size-dependent optical properties
 - Controlled geometries – e.g., efficient transport in oriented quantum rods, nanotubes
 - Controlled chemical functionalization for directed charge separation (surface chemistry)
- Novel effects? (multiple exciton generation, excitonic behavior)

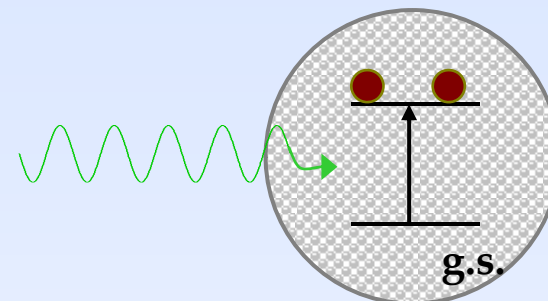
MEG



NCs of: IV-VI, II-VI, III-V, IV (e.g., PbSe, PbS, PbTe, CdSe, Si, InP), and **SWNTs** as well...

Spectroscopic Techniques: TA (interband, intraband), TRPL*, TRTS (THz), and quasi-cw PL*

(* shows red-shifted biexciton PL)



A. J. Nozik, *Physica E* **14**, 115 (2002).

R. Schaller and V. Klimov, *PRL* **92**, 186601 (2004).
(PbSe QDs \rightarrow QY 2.2)

Ellingson, Beard, Efros, Nozik, *et al.*, *Nano Lett.* **5**, 865 (2005).
(PbSe, PbS QDs \rightarrow QY 3.0)

R. Schaller *et al.*, *App. Phys. Lett.* **87**, 253102 (2005).
(CdSe \rightarrow QY 1.6 at $3.1E_g$)

J. Murphy *et al.*, *JACS* **128**, 3241 (2006).
(PbTe \rightarrow QY 3.0 at $4E_g$)

R. Schaller *et al.*, *Nano Lett.* **6**, 424-429 (2006).
(PbSe \rightarrow QY of 7.0 at $8E_g$)

J. Pijpers *et al.*, *J. Phys. Chem. C* **111**, 4146 (2007).
(InAs/CdSe/ZnSe QDs \rightarrow QY 1.6 at $2.7E_g$)

M. C. Beard *et al.*, *Nano Lett.* **7**, 2506 (2007).
(Si QDs \rightarrow QY 2.6 at $3.4E_g$)

R. Schaller *et al.*, *Nano Lett.* **7**, 3469 (2007).
(InAs/CdSe core/shell QDs \rightarrow QY of 1.95 at $\sim 4.8E_g$)

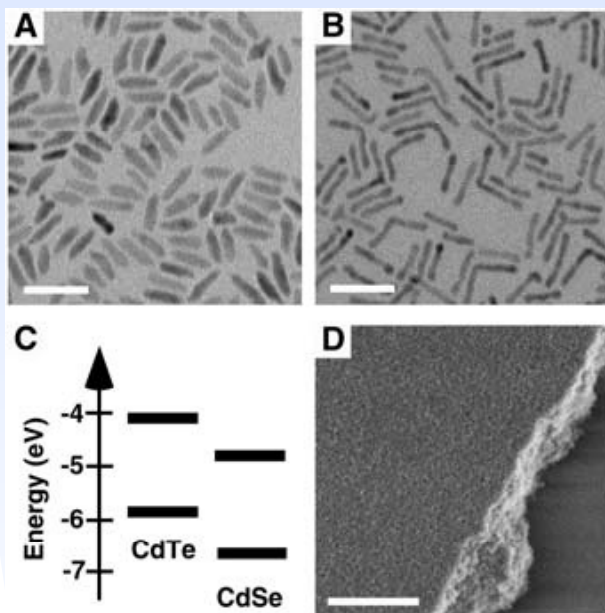
J. Luther *et al.*, *Nano Lett.* **7**, 1779 (2007).
(Efficient MEG in PbSe QD films)

M. Tuan Trinh *et al.*, *Nano Lett.* **8**, 1713 (2008).
(PbSe QDs \rightarrow QY 1.7 at $4.8E_g$)

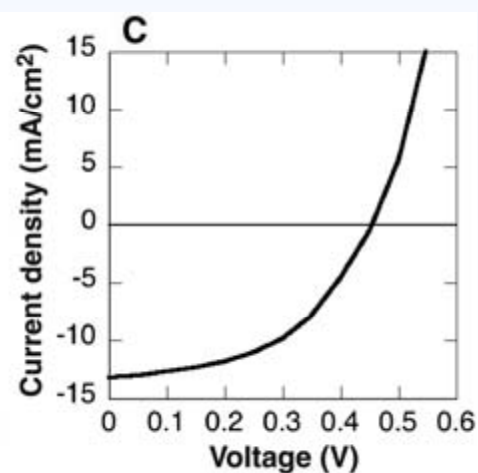
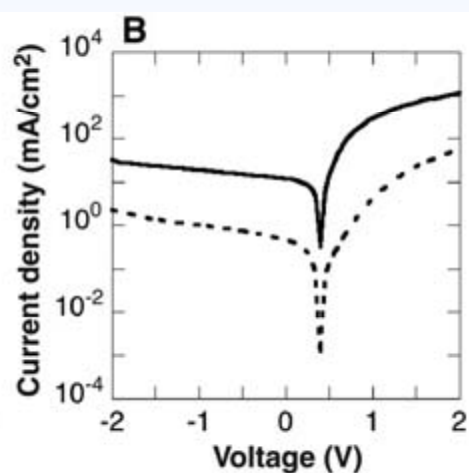
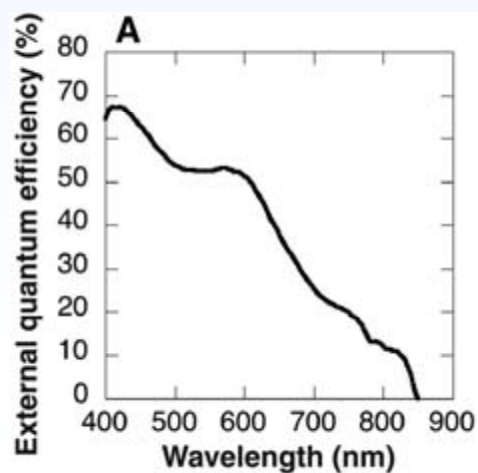
Air-Stable All-Inorganic Nanocrystal Solar Cells Processed from Solution

Ilan Gur, Neil A. Fromer, Michael L. Geier, A. Paul Alivisatos

21 OCTOBER 2005 VOL 310 SCIENCE



TEM images of (A) CdSe and (B) CdTe NCs. Scale bar, 40 nm. (C) An energy diagram of valence and conduction band levels for CdTe and CdSe illustrates the type II charge-transfer junction formed between the two materials...



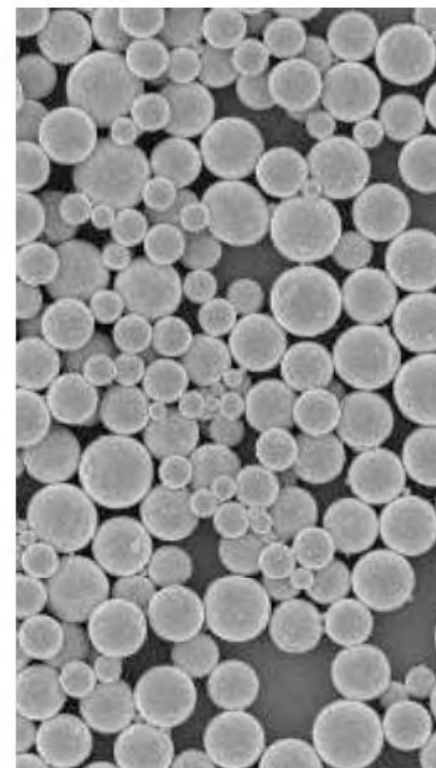


Figure 6 and 7: A laboratory sample of our nanoparticle ink. Nanoparticles shown to the right are an average of 20nm in diameter.

from NanoSolar white paper



Good solar cells from nanoparticle inks (CIGS)



Nanosolar CdS/Cu(In,Ga)Se₂ Cell

Device ID: H09B071-01C #2

Device Temperature: 24.0

Apr 09, 2009 12:31

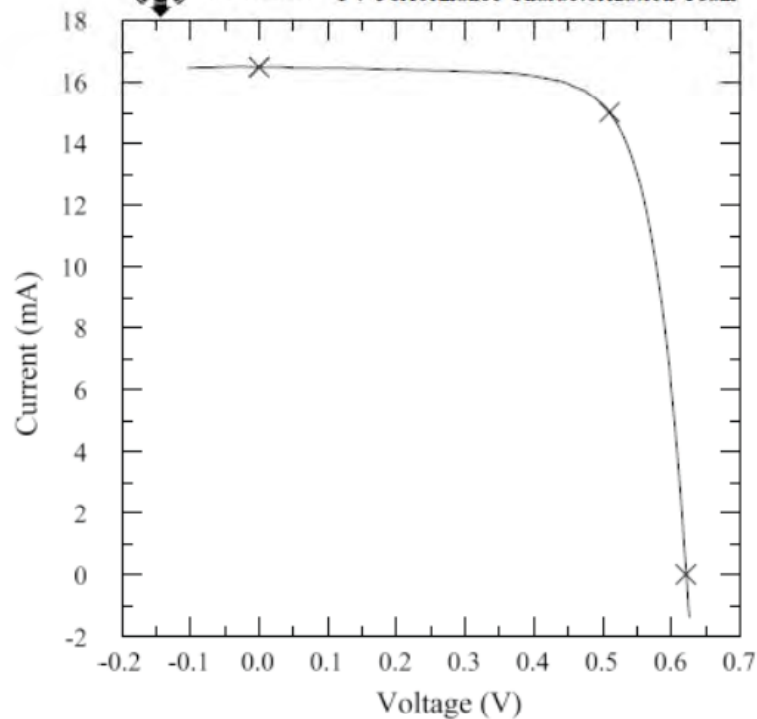
Device Area: 0.5000 cm²

Spectrum: ASTM G173 global

Irradiance: 1000.0 W/m²



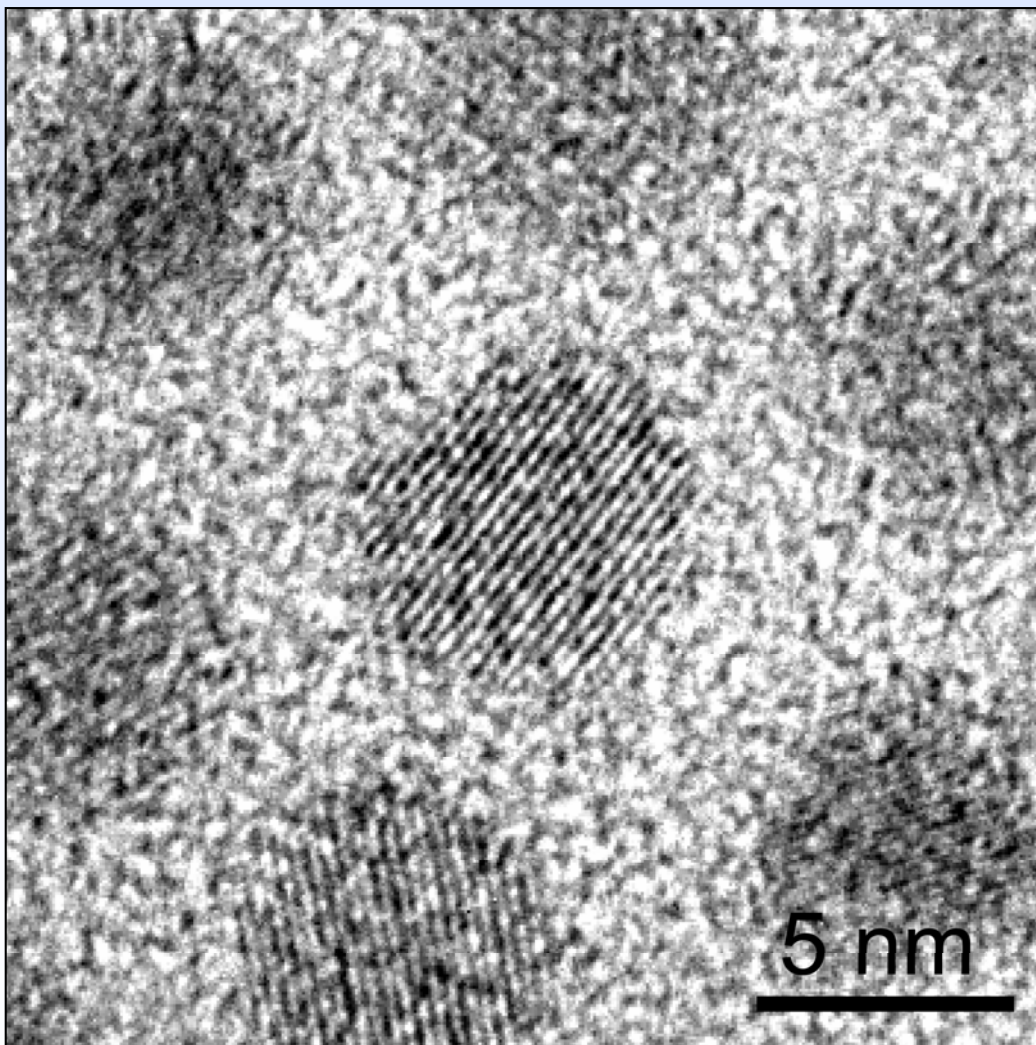
X25 IV System Confidential
PV Performance Characterization Team



$V_{oc} = 0.6214$ V
 $I_{sc} = 16.490$ mA
 $J_{sc} = 32.980$ mA/cm²
Fill Factor = 74.70 %

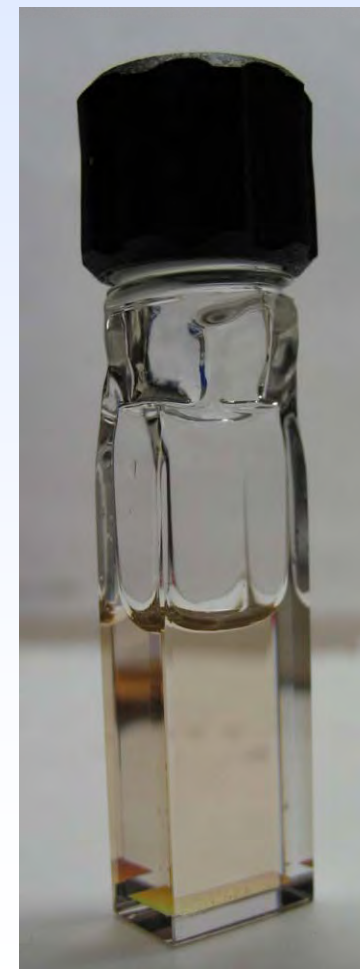
$I_{max} = 15.010$ mA
 $V_{max} = 0.5101$ V
 $P_{max} = 7.6540$ mW
Efficiency = 15.31 %

from NanoSolar white paper



PbSe exciton Bohr radius ~ 46 nm

PbSe QDs





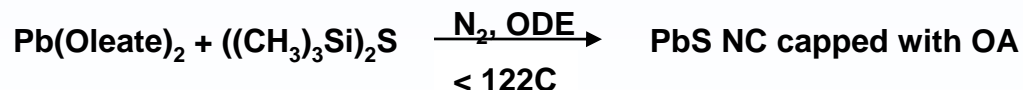
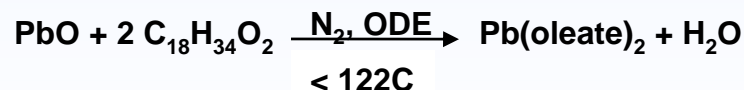
Synthesis of Colloidal PbS NCs

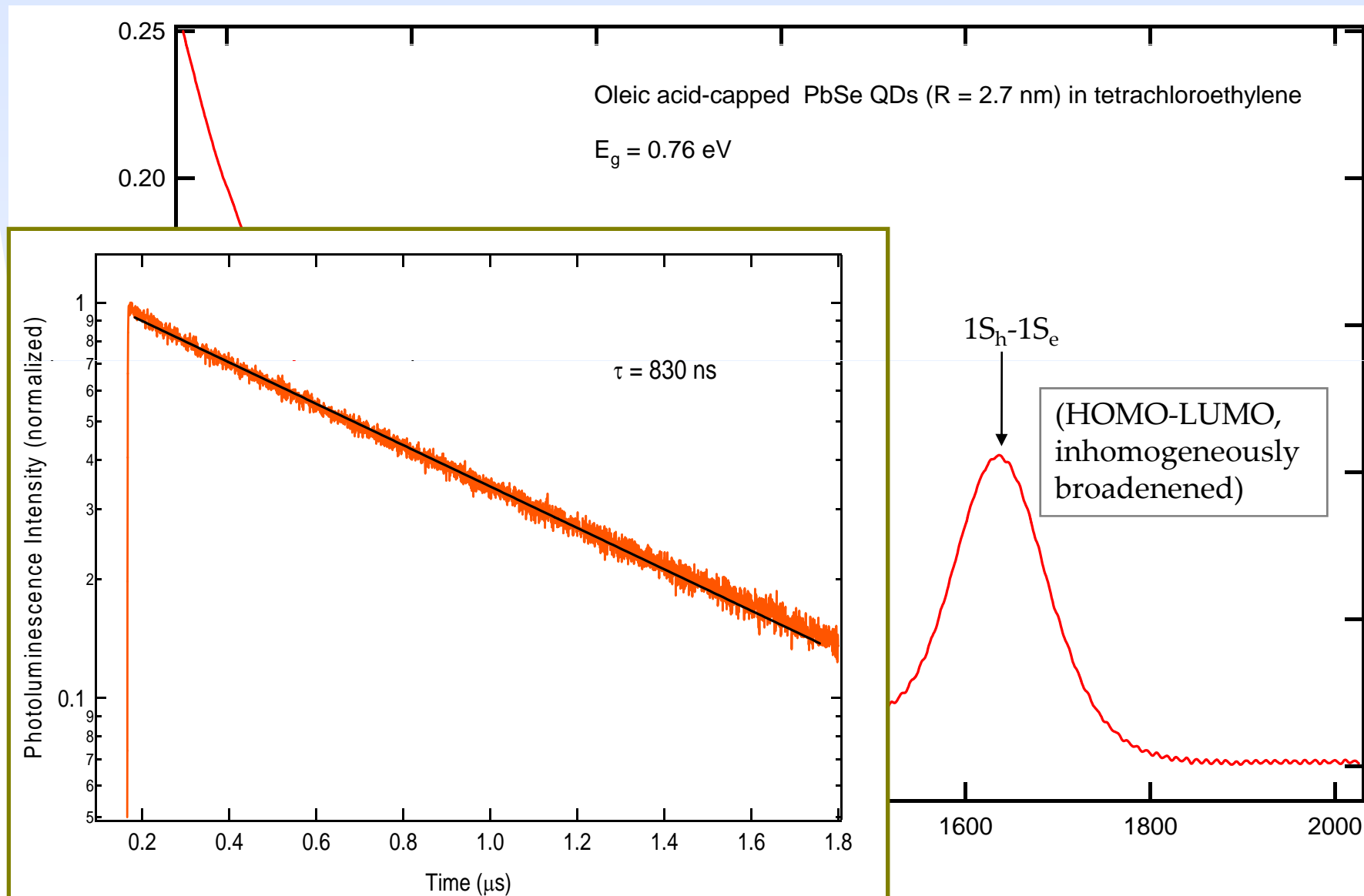


Hines, M. A.; Scholes, G. D. *Adv. Mater.* **2003**, *15*, 1844

Sample synthesis recipe (Krauss et al., adapted from Hines and Scholes)

- heat together in a flask at 150 °C for 1 hr under N₂ flow (to make lead oleate):
 - 220 mg lead(II) oxide
 - 0.25 ml oleic acid (tech grade)
 - 9.75 ml 1-octadecene (ODE, tech grade)
- after cooling to 90-100 °C, quickly inject into the reaction flask 5 ml of 0.1 M hexamethyldisilathiane (0.5 mmol) dissolved in ODE
- immediately cool with an ice bath
- precipitate (methanol/butanol mixture), centrifuge and resuspend in non-polar solvent (e.g., toluene, tetrachloroethylene, or hexanes).

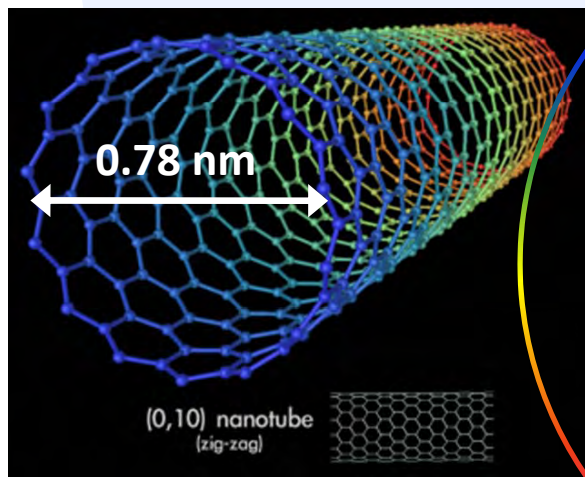




- Solution synthesis, typically at elevated temperature
- Inorganic crystalline core surrounded by organic capping molecules
- Most frequently suspended in organic solvents
- Can be stored stably either in powder or solution form

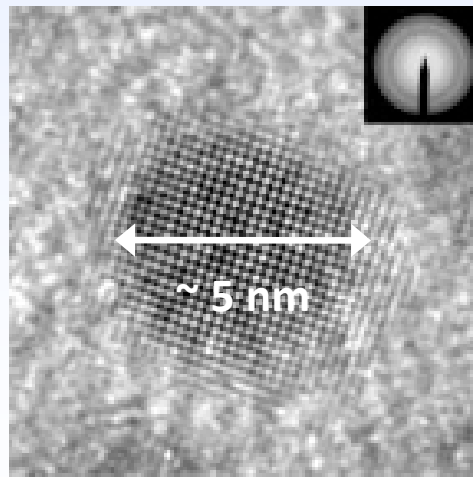
Colloidal semiconductor NCs have typical diameters of 1 nm to 10 nm

SWNT (10,0)



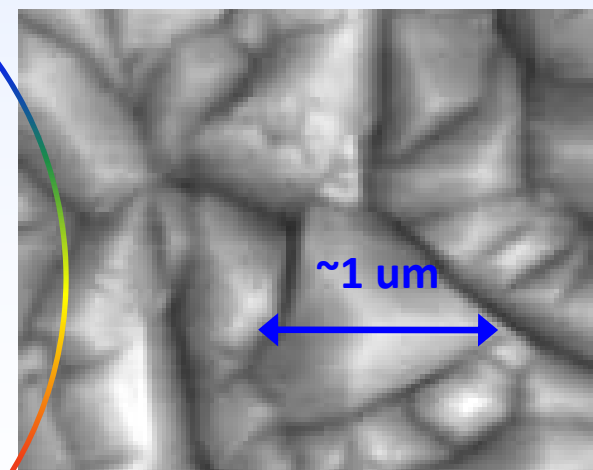
http://commons.wikimedia.org/wiki/Main_Page

PbS NC

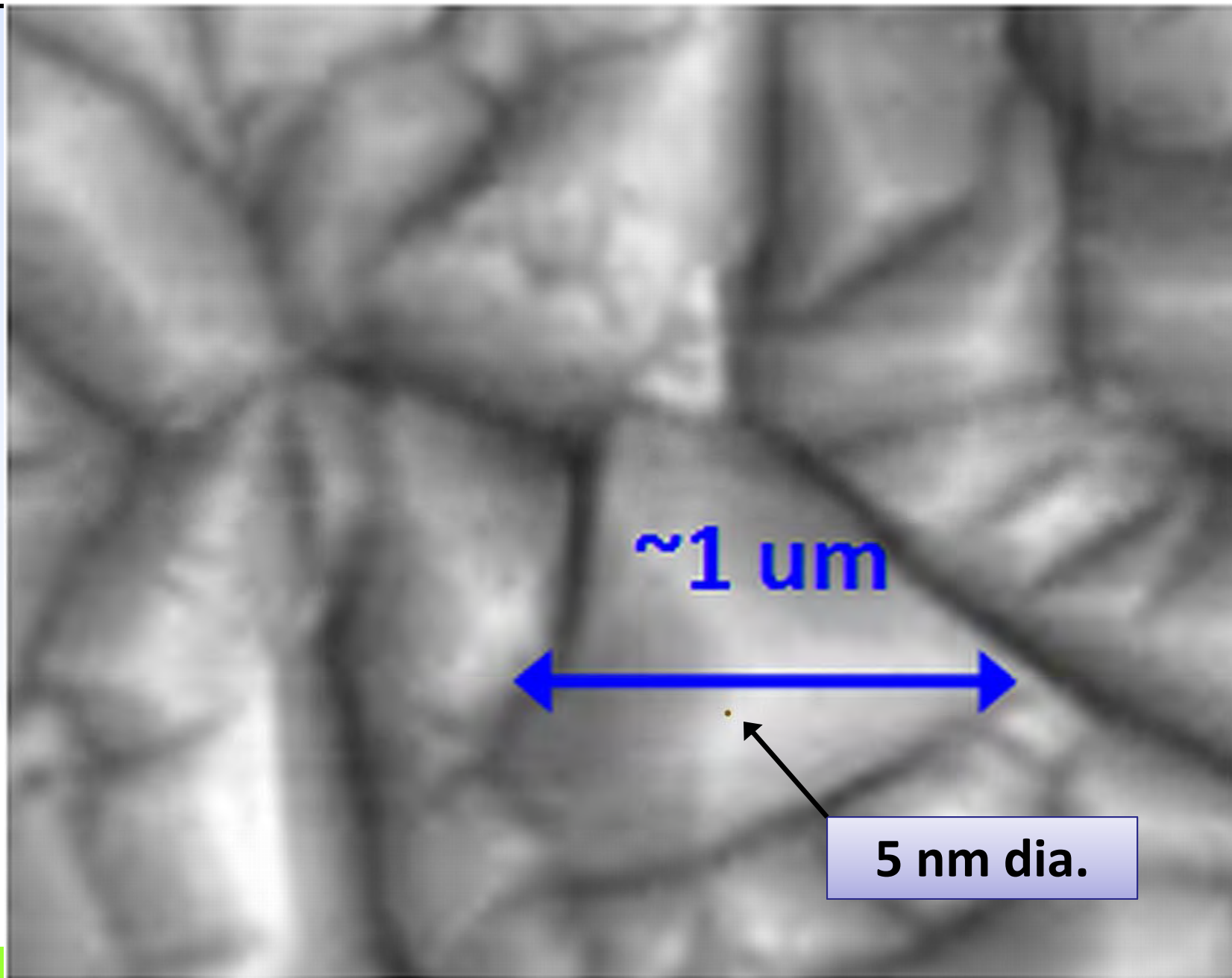


Sargent, *Adv. Mat.* (2005), 515-522.

CdTe grain size (CSS)



Quiñones et al., *J. Mater. Sci: Mat. Electron* (2007), 1085–1091



9794 *J. Phys. Chem. B, Vol. 102, No. 49, 1998*

Micic et al.

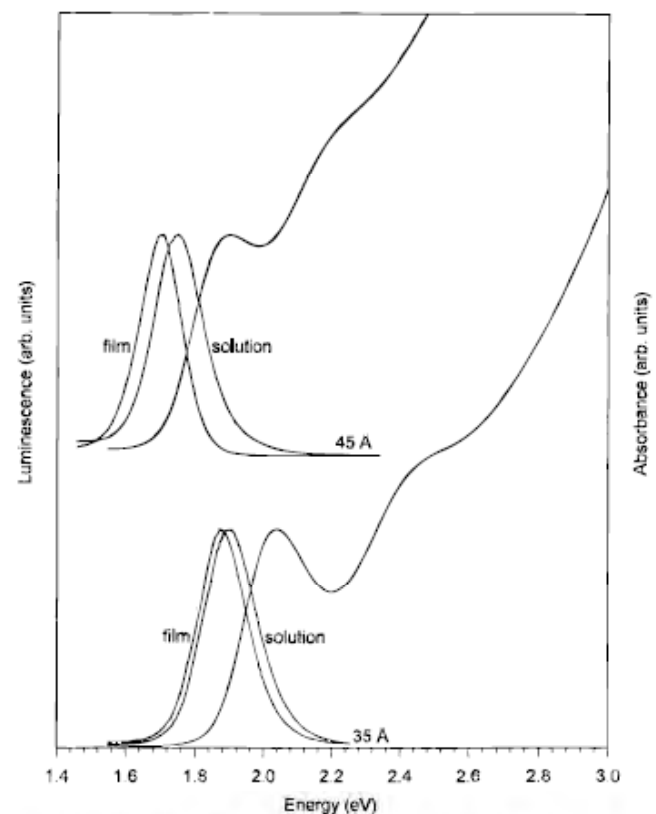
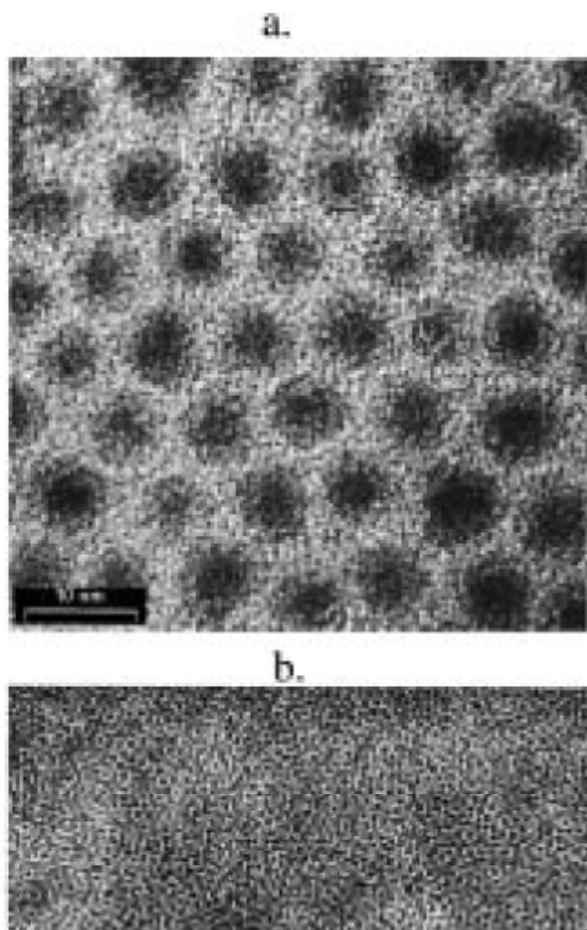


Figure 6. Absorption and luminescence spectra ($\lambda_{exc} = 500$ nm) of InP QDs in dilute colloidal solution and in close-packed films for 45- and 35-Å diameter QDs.

Optical, Electronic, and Structural Properties of Uncoupled and Close-Packed Arrays of InP Quantum Dots
 Olga I. Micic et al., 1998.



Coupling mechanisms



What could cause a decrease in QD-QD spacing to red-shift the exciton transition?

“This can cause (i) an increase in the average dielectric constant of the film, (ii) an increase in the inter-QD radiative coupling, and (iii) an increase in the inter-QD electronic coupling.”

Strong Electronic Coupling in Two-Dimensional Assemblies of Colloidal PbSe Quantum Dots, Williams et al., ACS Nano (2009)

- i) Increased wavefunction extension
- ii) Increased FRET
- iii) Reduced barrier width for tunneling/hopping transport

$$a_B^* = \frac{4\pi\epsilon\hbar^2}{m^*e^2}$$



Energy/charge transport mechanisms



Energy transfer (FRET):

$$E = \frac{1}{1 + \left(\frac{r}{R_0}\right)^6}$$

$\Rightarrow R_0 \approx 8-9$ nm for
 ~ 3 nm InP QDs

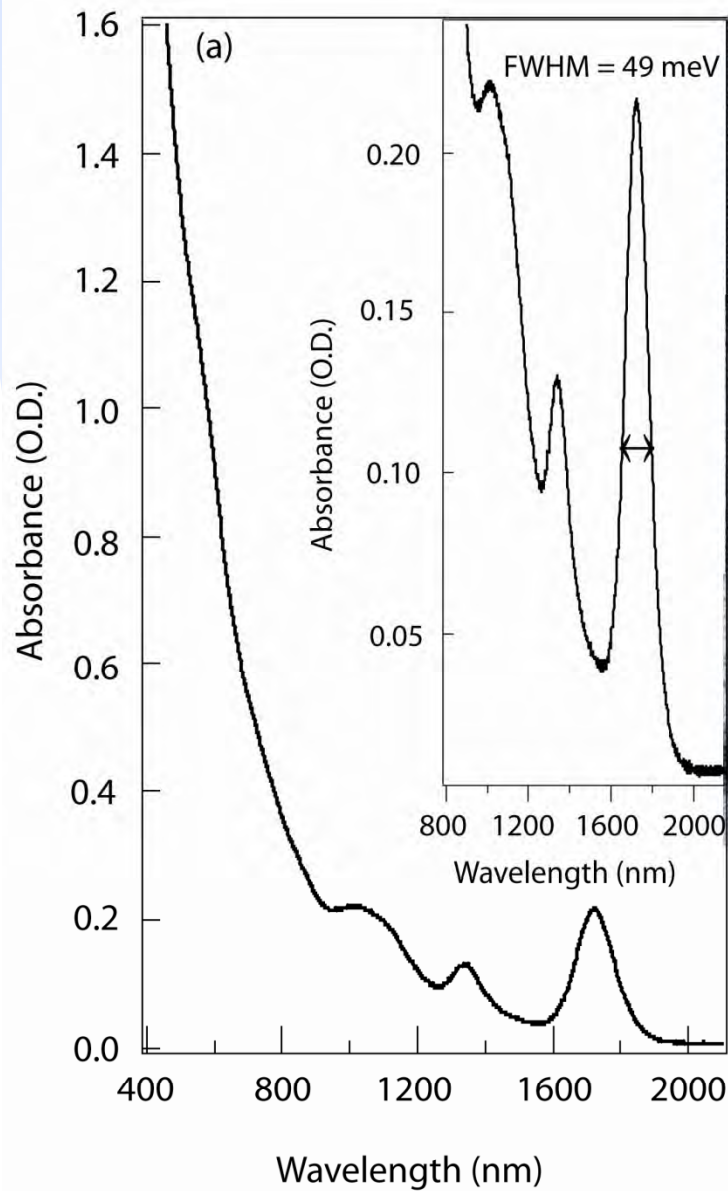
J. Phys. Chem. B **1998**, **102**, 9791-9796

Hopping/tunneling transport:

$$\Gamma_{ij} = \Gamma_0 \exp(-\beta d) \begin{cases} \exp\left[-\frac{E_j - E_i}{kT}\right] & E_j > E_i \\ 1 & E_i > E_j \end{cases}$$

Miller, Abrahams, *Phys. Rev.* **120**, 745–755 (1960)

Liu, Hillhouse, Law, *Nano Lett.* (2010), **10**, 1960.



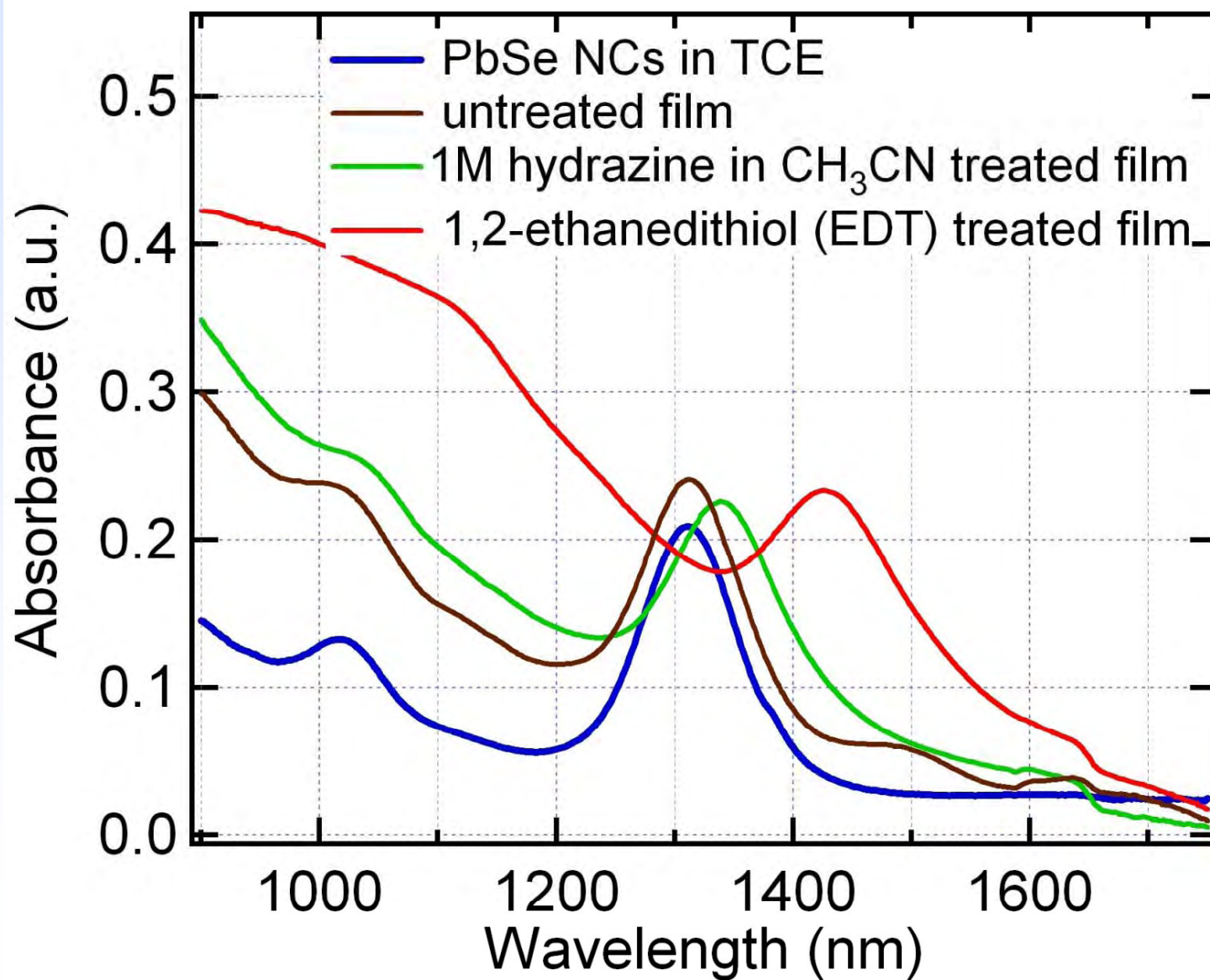
oleic acid (1.8 nm)

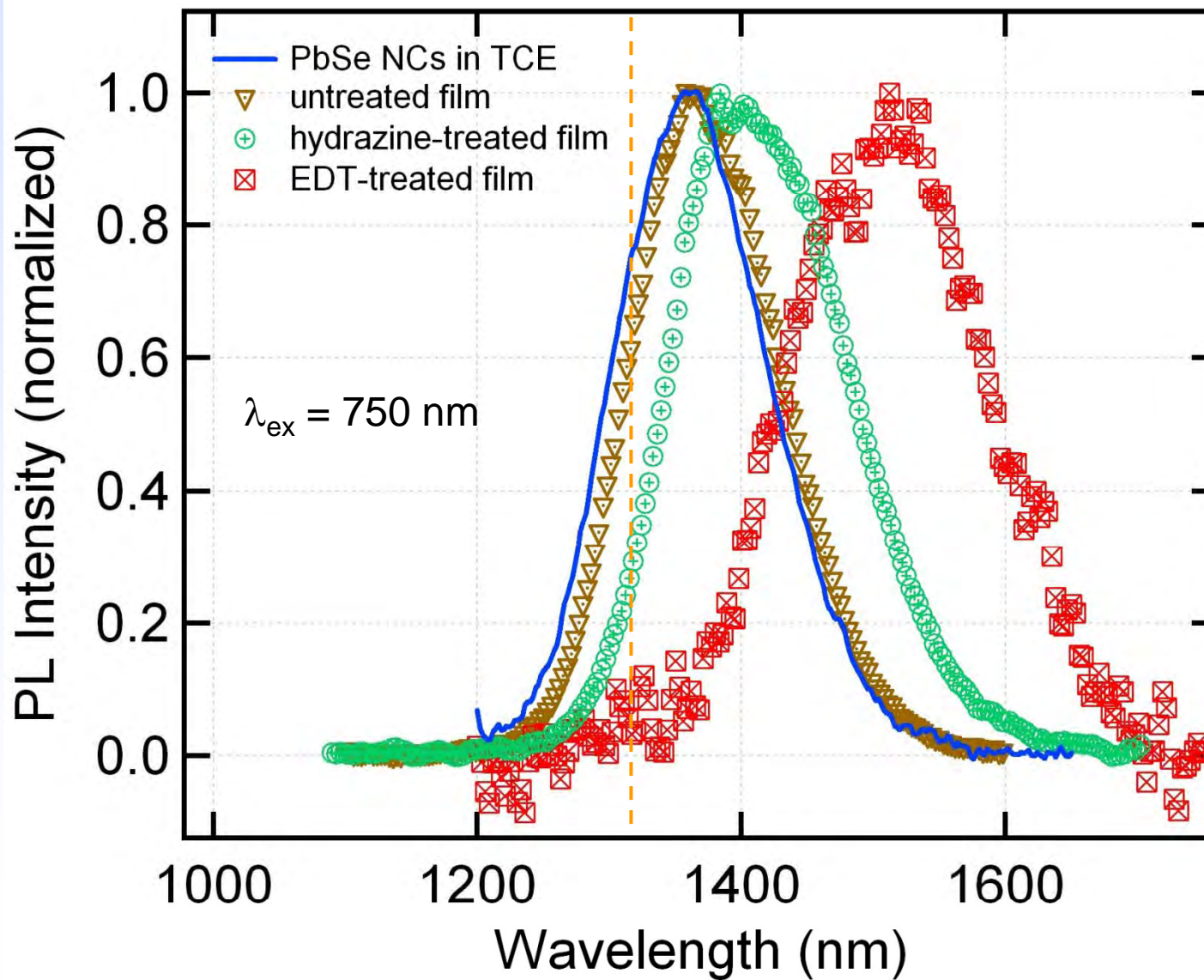
aniline (0.8 nm)

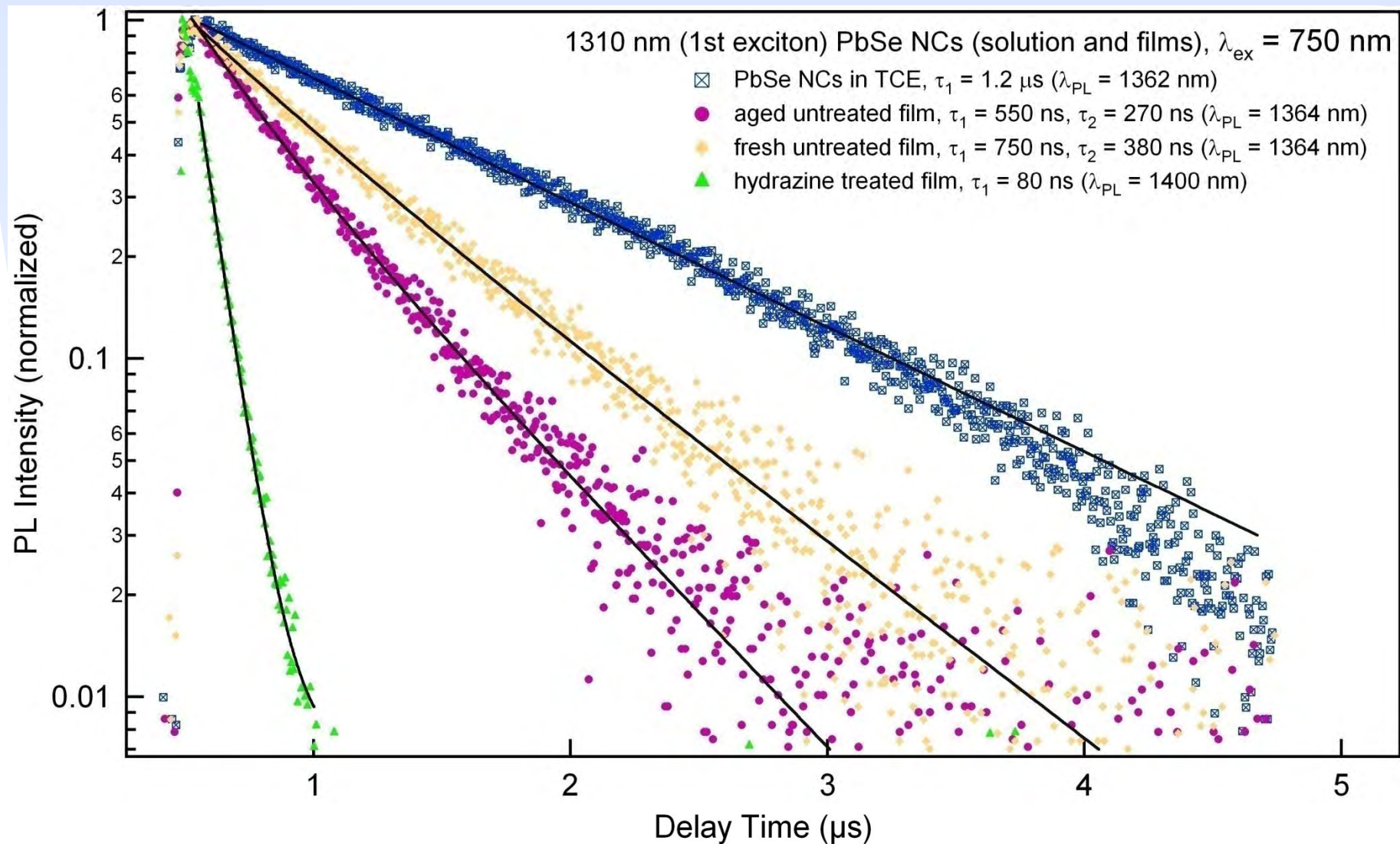
ethylenediamine (0.4 nm)

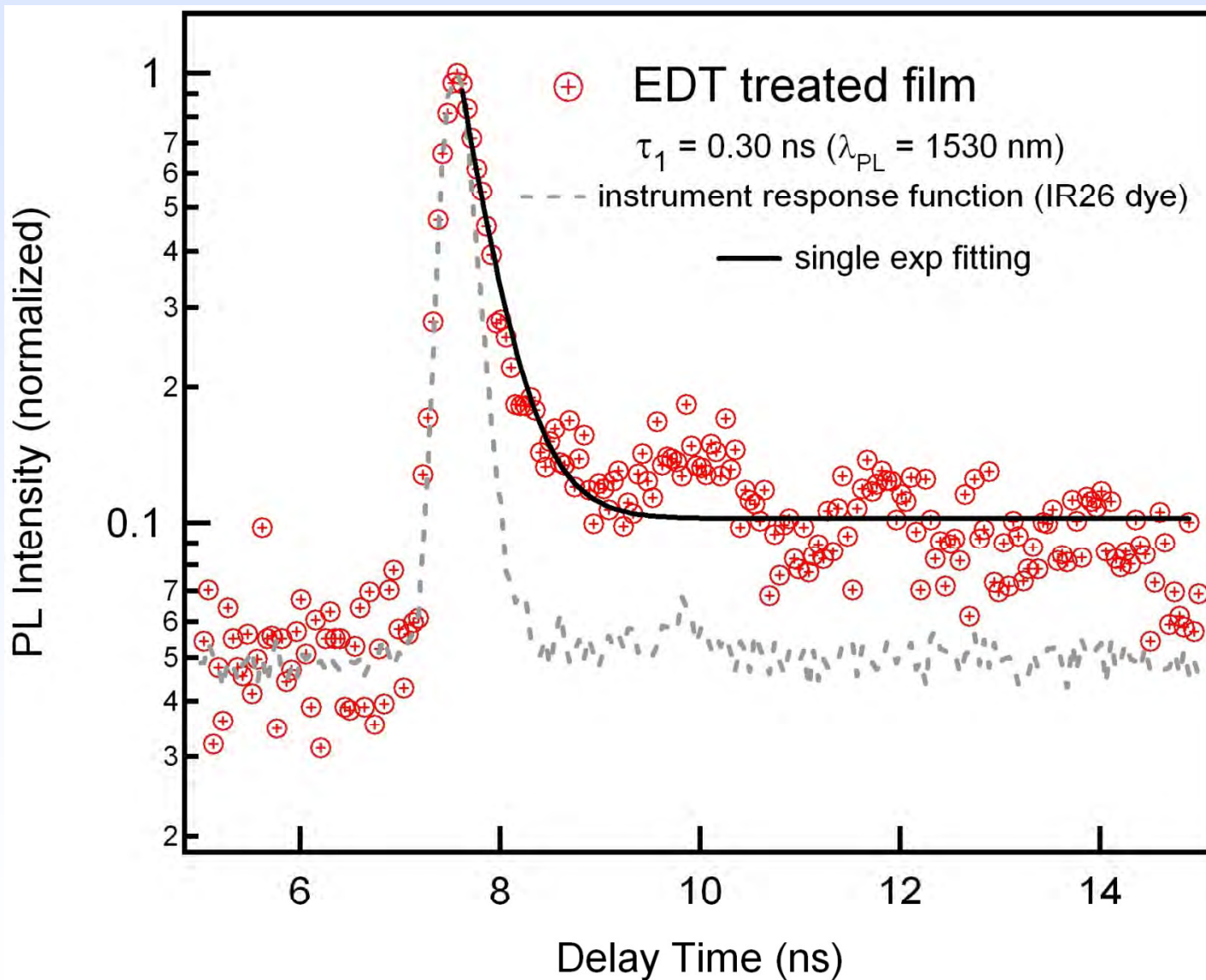
Artemyev *et al.*, J. Phys. Chem. B 2000, 11617.

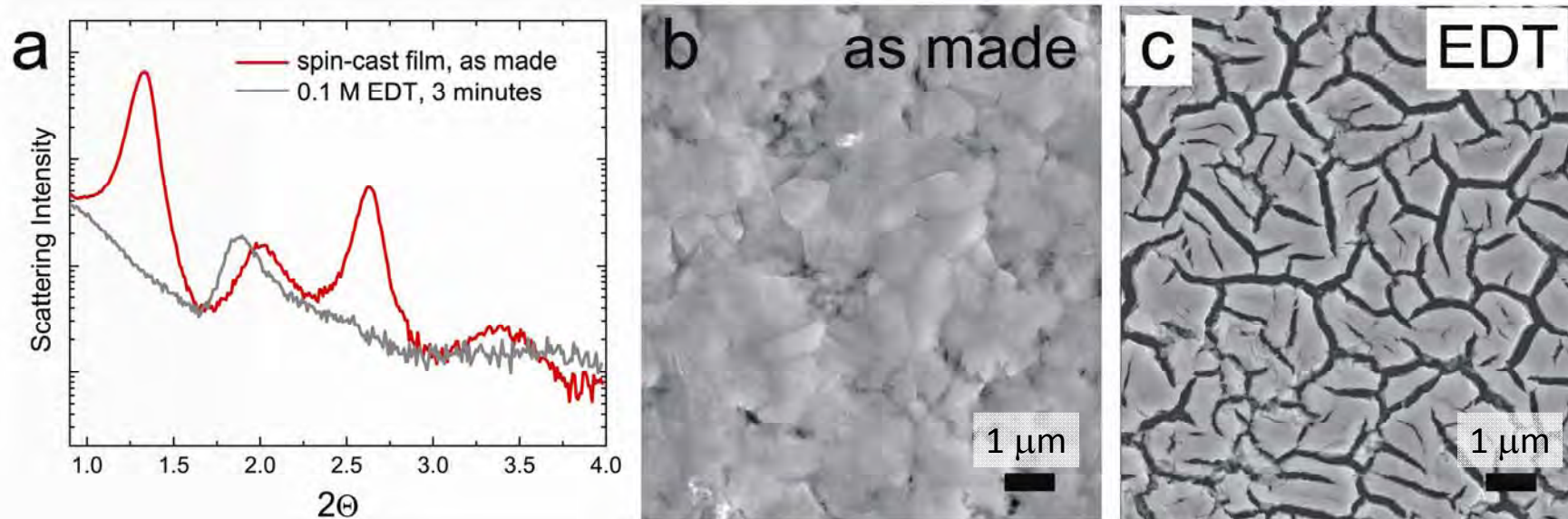
J. E. Murphy *et al.*, J. Phys. Chem. B 2006, 25455.





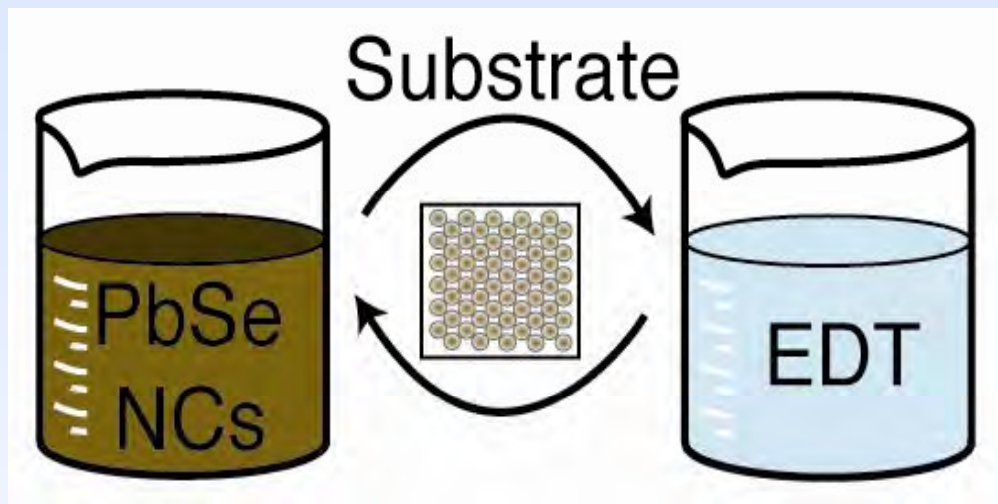






“Microstructure of the spin-cast NC films before and after EDT treatment. (a) SAXS data, showing a ~ 16 Å decrease in the spacing between the NCs and a dramatic loss of superlattice order upon EDT treatment. Measurements were taken in air. (b, c) Plan-view SEM images of (b) an untreated film and (c) a treated film.”

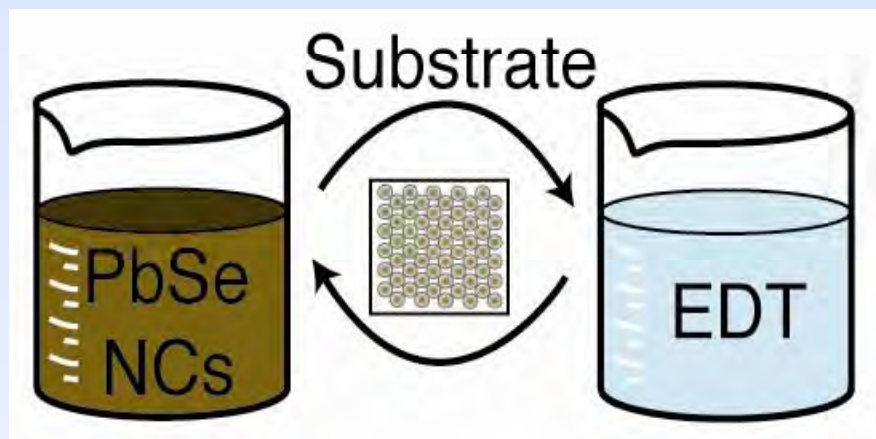
J. Luther, M. Law et al., ACS Nano 2, 271 (2008).



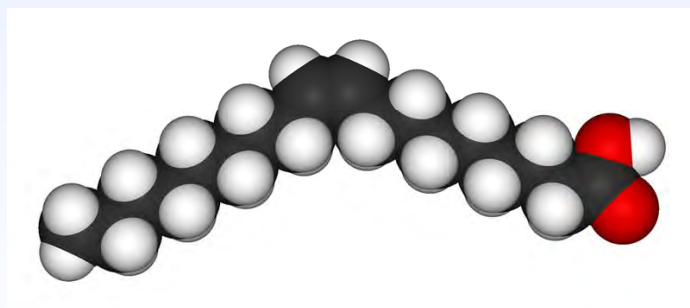
EDT = 1,2-ethanedithiol

Layer by layer (LbL) fabrication of PbSe nanocrystal (NC) films. Nanocrystal films prepared by dip-coating, alternating between (1) PbSe NCs in hexane and (2) 0.1 M EDT in anhydrous acetonitrile, allowing the film to dry between each layer.

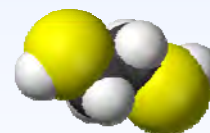
J. M. Luther, M. Law *et al.*, "Structural, Optical, and Electrical Properties of Self-Assembled Films of PbSe Nanocrystals Treated with 1,2-Ethanedithiol", *ACS Nano* **2**, 271 (2008).



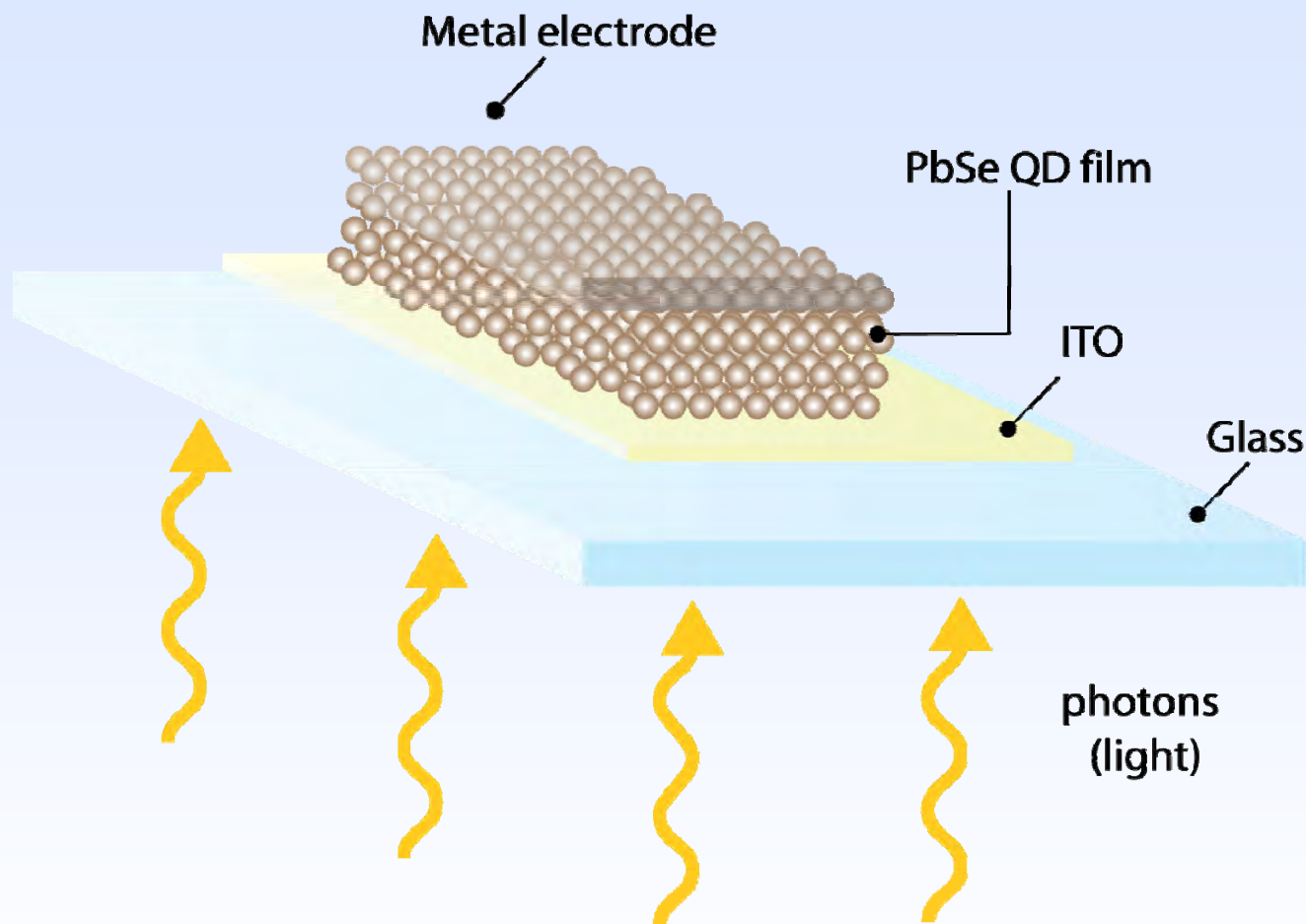
EDT = 1,2-ethanedithiol



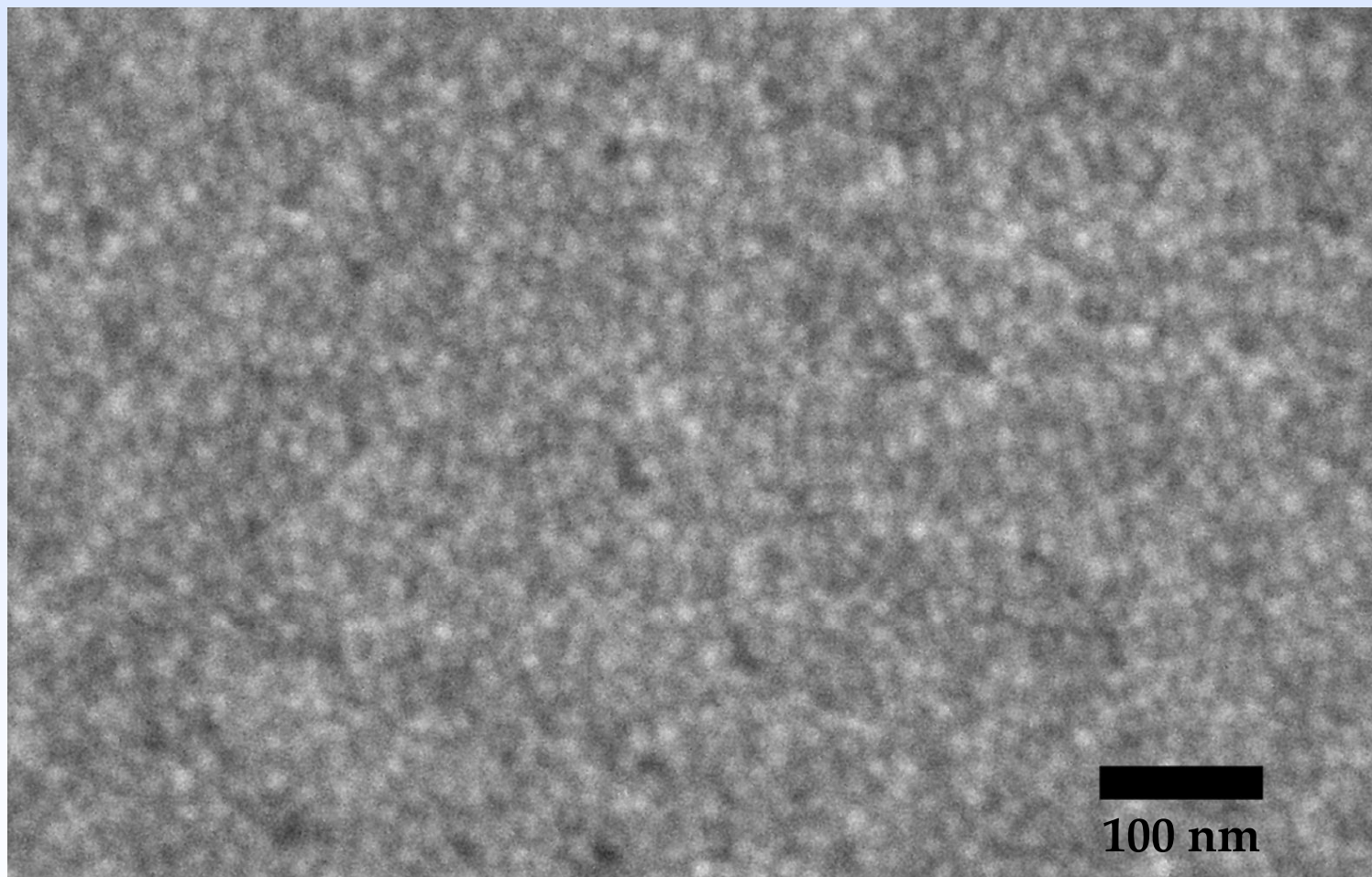
Oleic acid, $C_{18}H_{34}O_2$
 $L \approx 20 \text{ \AA}$



1,2-ethanedithiol
 $C_2H_6S_2$
 $L \approx 6 \text{ \AA}$

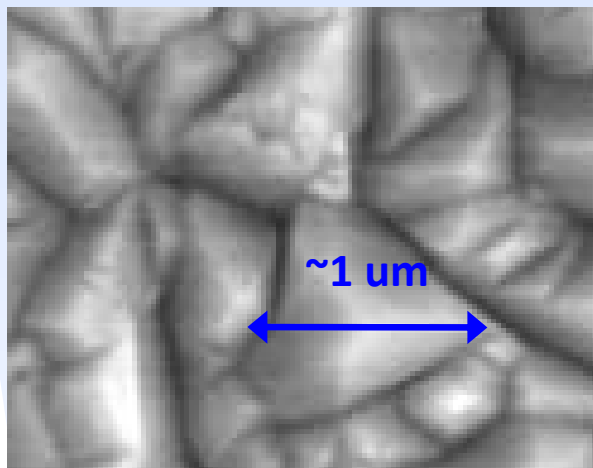


- Auger recombination typically occurs on $\lesssim 100$ ps timescale (need fast exciton dissociation and/or diffusion to low density)
- High long-range mobility desired
- EQE > 100% would provide MEG process/concept confirmation

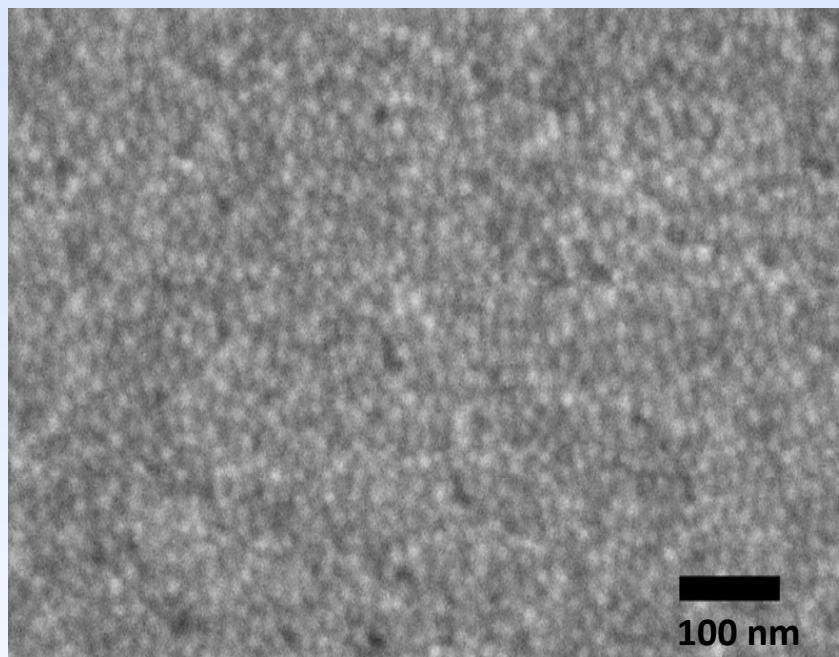


Plan view SEM image of PbSe NC films prepared by layer-by-layer dip coating onto ITO substrate. LbL film shown was produced using 10-20 dip coating cycles.

J. Luther, M. Law *et al.*, *ACS Nano* **2**, 271 (2008).



e.g. CdTe



Rough estimates of grain boundary densities (boundary area / layer area)

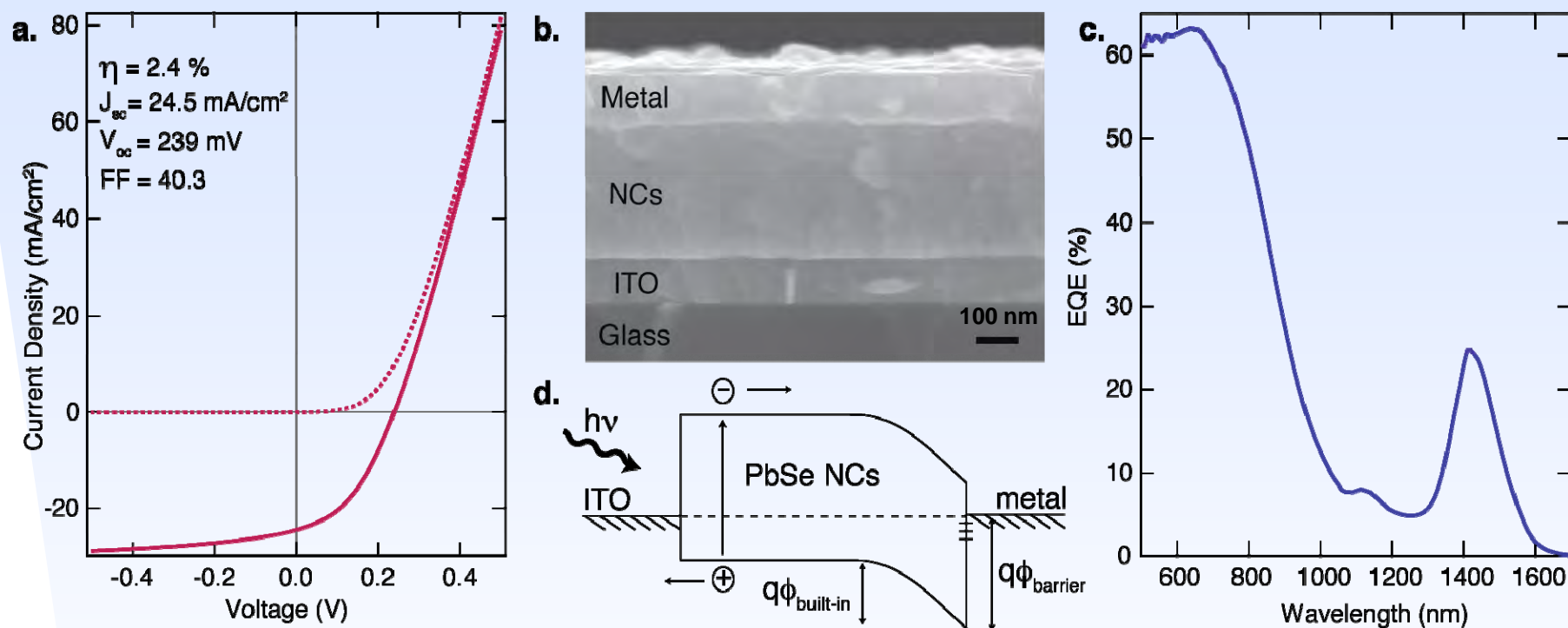
CdTe: $\sim 1 \mu\text{m}$ grain size (assumed cubic grains, $\sim 1 \mu\text{m}$ thick layer):

$$3 (\mu\text{m}^2 / \mu\text{m}^2)$$

PbSe QD film: $\sim 5 \text{ nm}$ “grain size” (assumed cubic grains 5 nm on a side, $1 \mu\text{m}$ thick film):

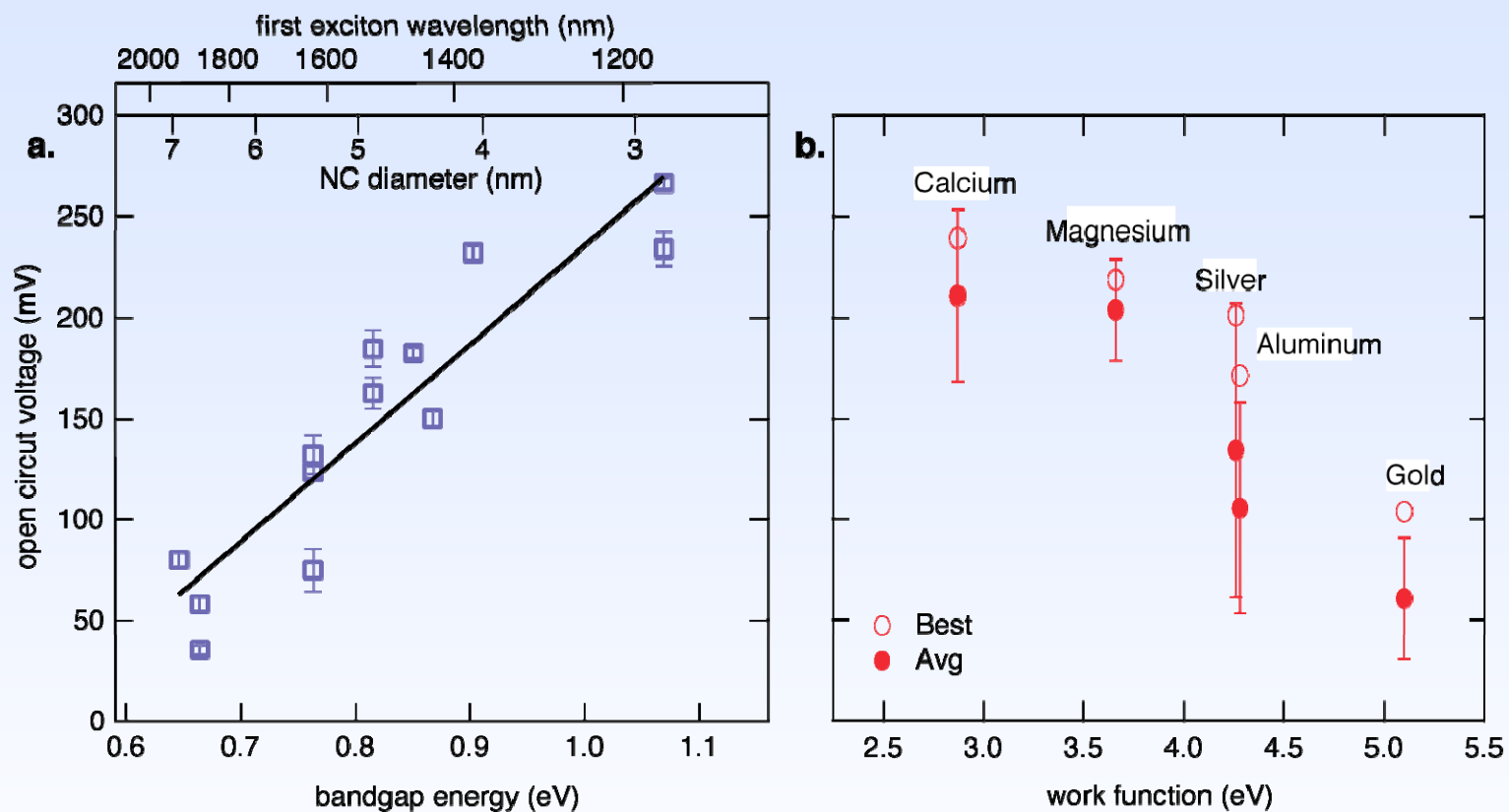
$$600 (\mu\text{m}^2 / \mu\text{m}^2)$$

$$S/V \propto 1/r$$



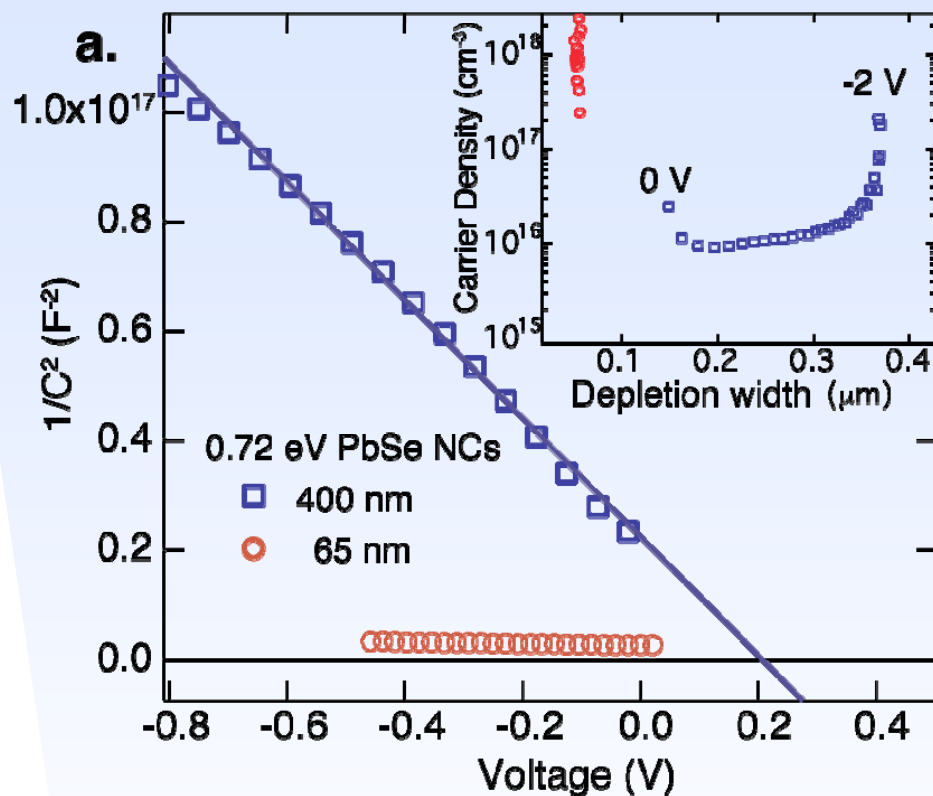
(a) J-V in dark and under 100 mW cm⁻² AM 1.5G ($E_g = 0.9$ eV). **(b)** SEM cross-section of the stack (metal is 20 nm Ca / 100 nm Al). **(c)** EQE of a device with 140 nm-thick NC film and an efficiency of $\sim 2.2\%$ ($E_g = 0.95$ eV). **(d)** Proposed equilibrium band diagram; band bending occurs at interface between the NCs and metal electrode. Schottky barrier height ($q\phi_{barrier}$) for p-type material is the difference between the valence band max. and the metal Fermi level.

J. M. Luther, M. Law, *et al.*, *Nano Lett.* **8**, 3488-3492 (2008).



(a) Dependence on NC size. Each point is the average of six devices from a single substrate. The line is a least-squares fit to the data. **(b)** Dependence on metal work function ($E_g = 0.82$ eV). The best and average V_{oc} for each metal are shown, along with the standard deviation of six devices.

J. M. Luther, M. Law, *et al.*, *Nano Lett.* **8**, 3488-3492 (2008).



Relation between doping density and QD density? QD size ~ 6 nm dia., add 1 ligand ~ 0.5 nm, so $V(QD) = (4\pi/3)(3.2 \text{ nm})^3 = 1.3 \times 10^{-19} \text{ cm}^3$

Random packing of spheres $\rightarrow \sim 0.64$

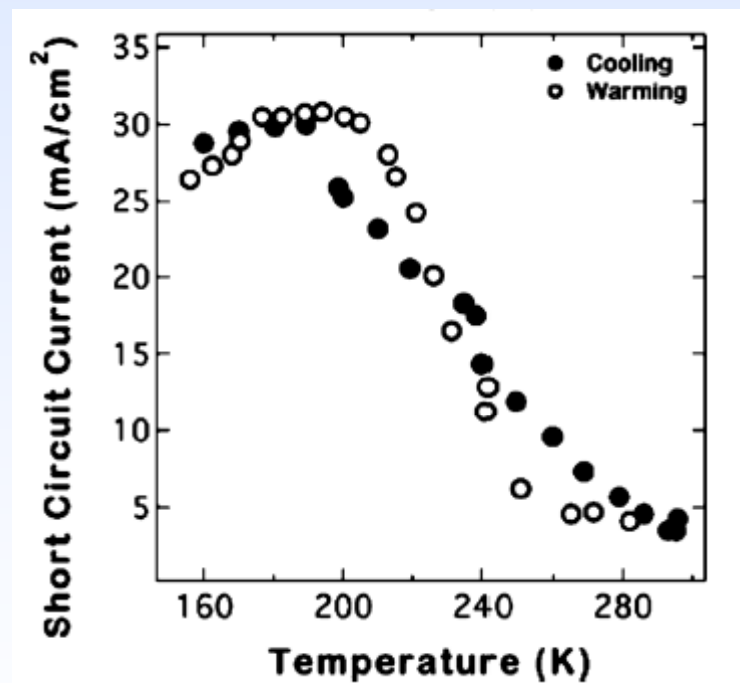
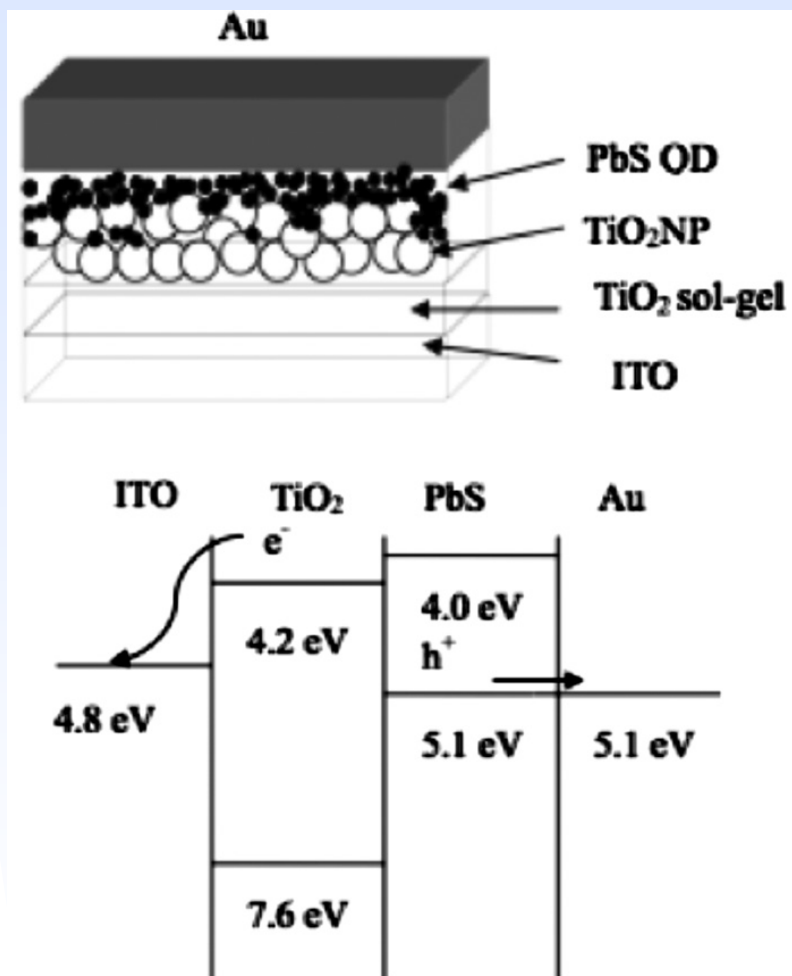
\rightarrow Vol. density of QDs $\sim 5 \times 10^{18} \text{ cm}^{-3}$

\rightarrow **~ 1 free carrier per 500 QDs**

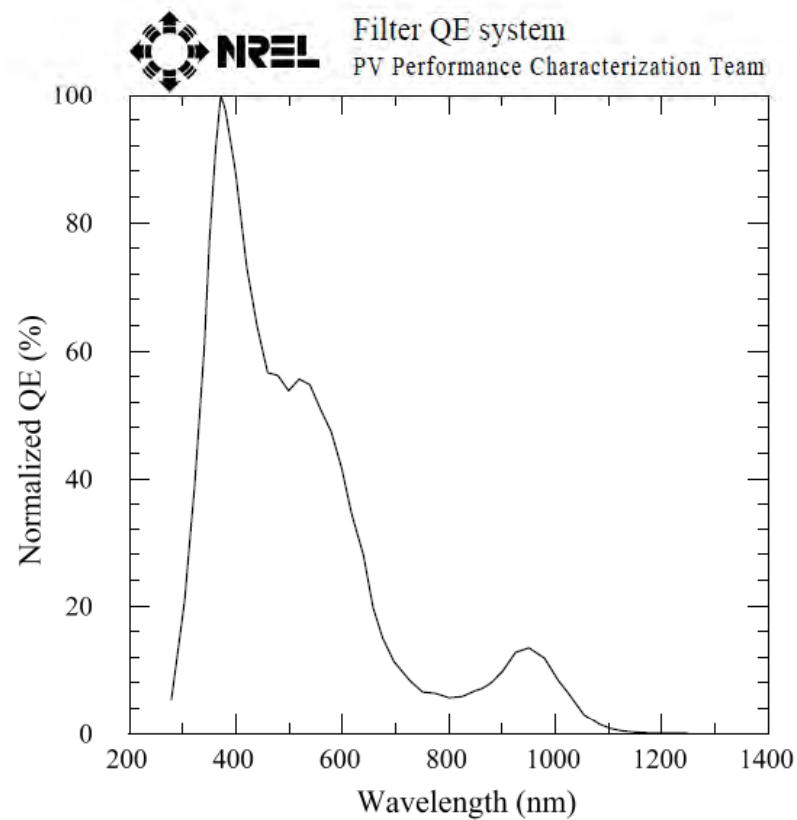
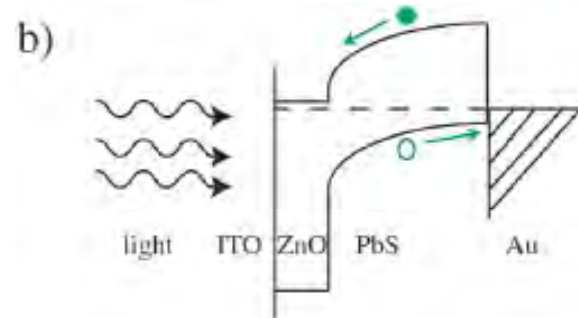
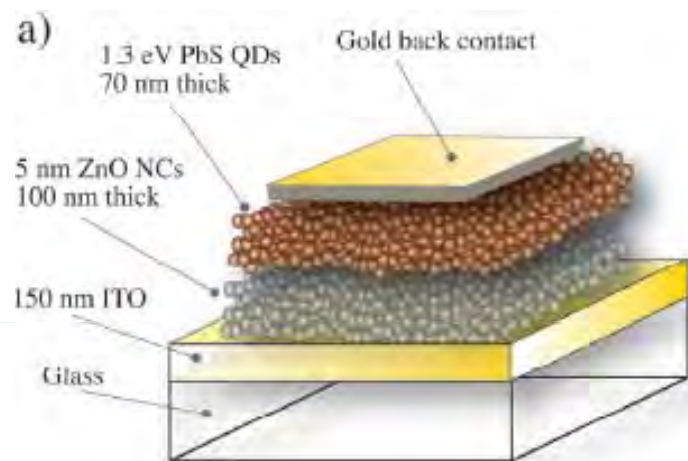
How/why?

“Mott-Schottky plots at 1 kHz for devices with a thin (65 nm, red) and thick (400 nm, blue) NC layer. The thick device has an equilibrium depletion width of ~ 150 nm, while the thin device is fully depleted.”

J. M. Luther, M. Law, *et al.*, *Nano Lett.* **8**, 3488-3492 (2008).



T. Ju et al., *APL* 97, 043106 2010

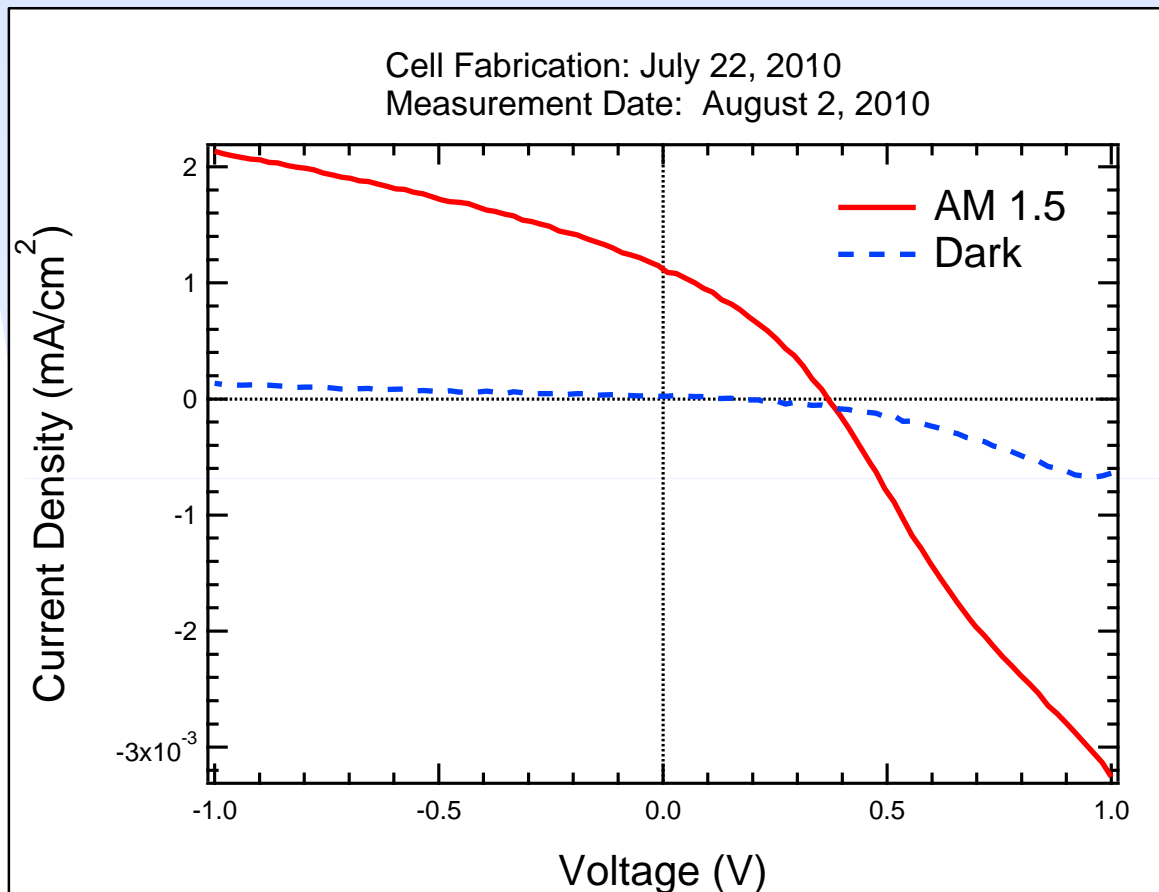


Luther et al., *Adv. Mater.* **2010**, *22*, 3704–3707

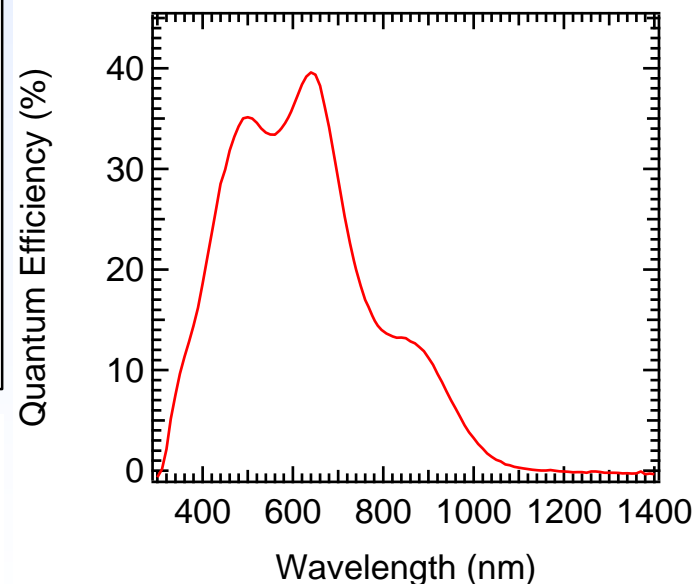
Layers of ZnO and PbS fabricated in air. Device shows good air stability over ~1,000 hrs.



PbS/ZnO heterojunction solar cell (UT)



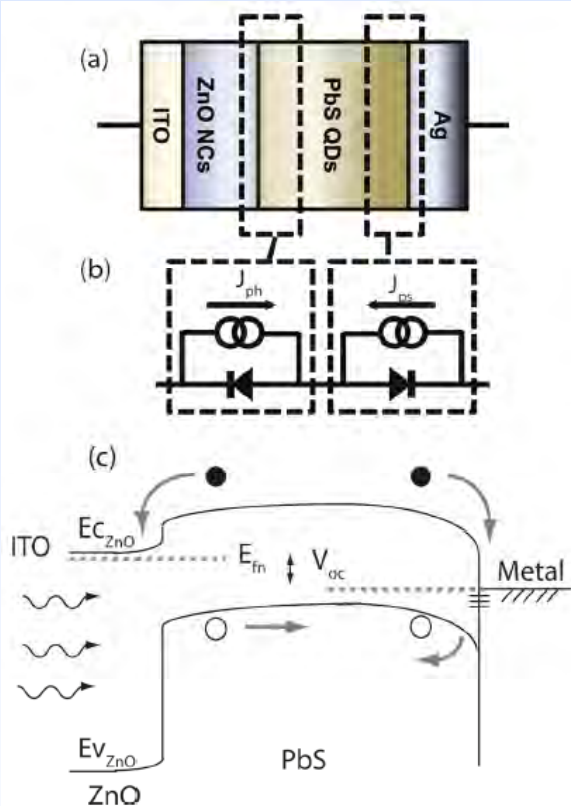
Silver back contact (150 nm)
PbS NCTF (Layer-by-Layer, ~120 nm)
ZnO (sputtered, ~150 nm)
Patterned ITO (150 nm)
glass



4.1 mg/mL concentration of PbS 820nm NC in hexanes;
1 mM solution of 1,2-ethanedithiol in acetonitrile
Layer-by-Layer deposition for 20 cycles (~ 120nm PbS film thickness)

$V_{OC} = 368$ mV $J_{SC} = 12.44$ mA/cm² FF = 0.33
Efficiency = 1.53% $R_{sh} = 613$ Ohms $R_s = 191$ Ohms





Two-diode model equivalent circuit including a main diode between ZnO QD film with PbS QD film, and a Schottky-diode between PbS QD film with top contact Au or Ag.

“Quantum Dot Size Dependent J-V Characteristics in Heterojunction ZnO/PbS Quantum Dot Solar Cells”, J. Gao et al, Nano Lett. 2011.

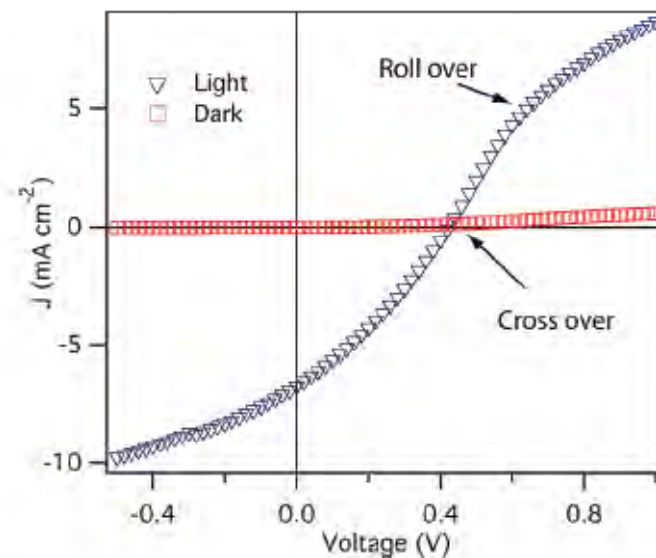
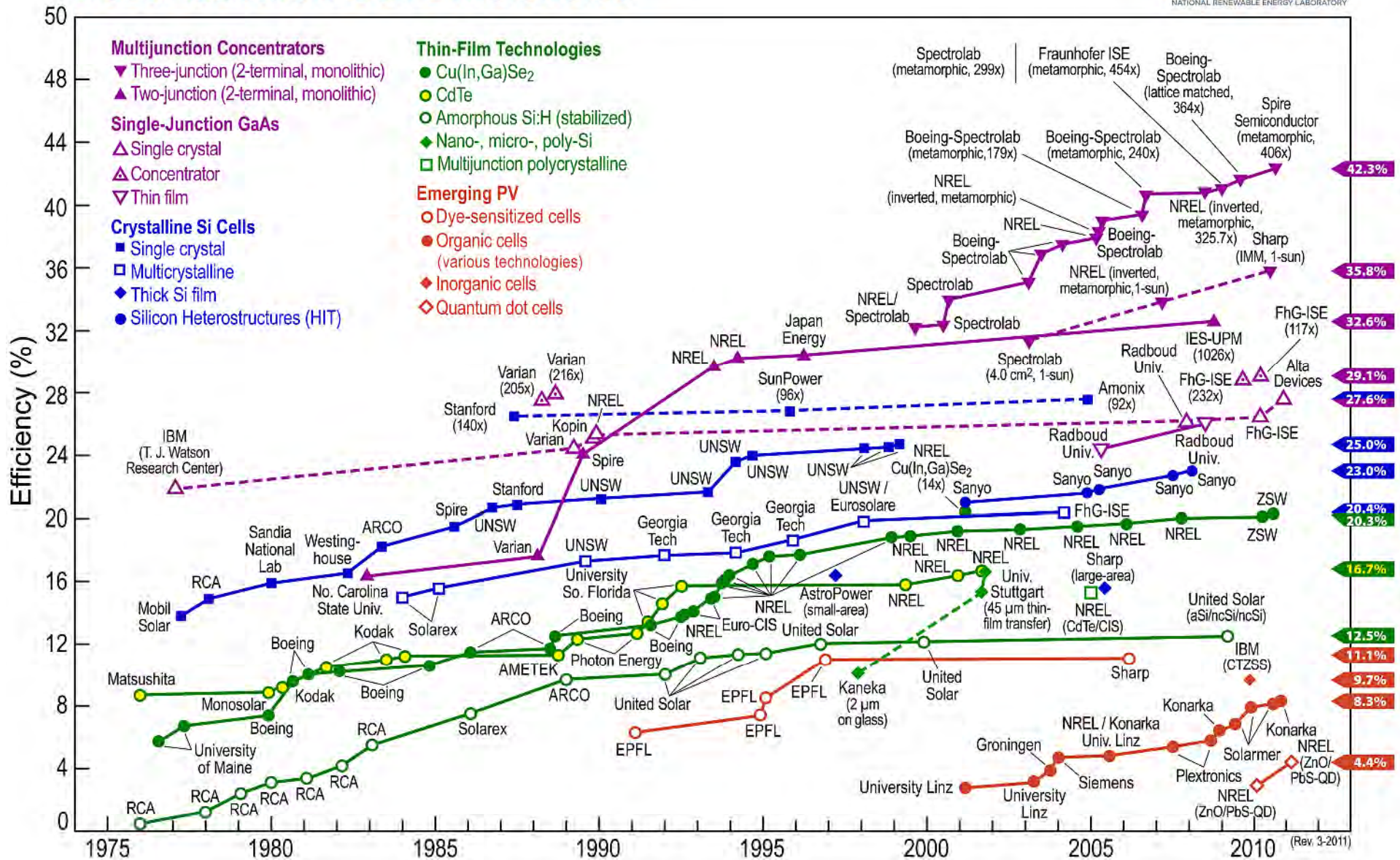


Figure 1. Typical characteristics of $J-V$ in the dark and under 1 sun illumination, which shows the crossover and roll-over effects. The device structure is ITO/ZnO NC/PbS QD/Ag with PbS layer thickness of ~ 550 nm.

1. **Roll-over effect**, a fact of current limiting at positive voltage, is due to a hole transport barrier at Schottky-junction formed by PbS QD film with metal contact .
2. **Cross-over effect**: low forward current due to Schottky barrier at metal contact (hole transfer barrier).

Best Research-Cell Efficiencies





Issues of interest for QD-based thin film PV



- Energy intensity of materials and deposition
- Doping control within films
- Air/moisture tolerance (deposition and operation)





Contributors/Collaborators



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Jianbo Gao (UT-PVIC postdoc @ NREL)
Khagendra Bhandari, NC thin films
Neale Haugen, laser spectroscopy
Paul Roland, device characterization and spectroscopic ellipsometry
Ryan Zeller, laser scribing
Tyler Kinner (undergraduate), Earth-abundant NC synthesis

Current Funding:

- UT-PVIC startup funds
- National Renewable Energy Lab/DOE-EERE
- AFRL



Left to right: Julia Deitz, Ryan Zeller, Neale 'Ole' Haugen, Khagendra Bhandari, Paul Roland, Tieneke Dykstra, and Randy Ellingson