X-ray Diffraction and Crystal Structures

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X-Ray Generation

X-rays are electromagnetic radiation with wavelength ~1 Å = 10^{-10} m (visible light ~5.5x10⁻⁷ m)

X-ray generation: electrons are emitted from the cathode and accelerated toward the anode. Here, Bremsstralung radiation occurs as a result of the "braking" process – X-ray photons are emitted.



X-Ray Generation

The most common metal used is copper, which can be kept cool easily, due to its high thermal conductivity, and which produces strong K_{α} and K_{β} lines. The K_{β} line is sometimes suppressed with a thin (~10 µm) nickel foil.

- K-alpha (K_{α}) emission lines result when an electron transitions to the innermost "K" shell (principal quantum number 1) from a 2p orbital of the second or "L" shell (with principal quantum number 2).
- The K_{α} line is actually a doublet, with slightly different energies depending on spin-orbit interaction energy between the electron spin and the orbital momentum of the 2p orbital.

 $\lambda(K_{\alpha}) = 0.154 \text{ nm}$ $\lambda(K_{\beta}) = 0.139 \text{ nm}$

from http://en.wikipedia.org/wiki/K-alpha



Atomic levels involved in copper K_{α} and K_{β} emission.

$K_{\alpha}\,and\,K_{\beta}\,X\mbox{-ray lines}$



from Preston and Dietz, p. 191.

X-Ray diffraction





X-Ray Diffraction -- Bragg's Law

Diffraction of x-rays by crystal: spacing *d* of adjacent crystal planes on the order of 0.1 nm

→ three-dimensional diffraction grating with diffraction maxima along angles where reflections from different planes interfere constructively

2*d* sin θ = $m\lambda$ for m = 0, 1, 2, ...

Note that your measured XRD spectra will most likely reveal only 1^{st} order diffracted lines (i.e., those for which m = 1).

The Braggs (Bragg's Law)





Sir William Henry Bragg 1862-1942

William Lawrence Bragg 1890-1971

Bragg occupied the Cavendish chair of physics at the University of Leeds from 1909. He continued his work on X-rays with much success. He invented the X-ray spectrometer and with his son, William Lawrence Bragg, then a research student at Cambridge, founded the new science of X-ray analysis of crystal structure.

In 1915 father and son were jointly awarded the Nobel Prize in Physics for their studies, using the X-ray spectrometer, of X-ray spectra, X-ray diffraction, and of crystal structure.

http://en.wikipedia.org/wiki/William_Henry_Bragg

X-Ray Diffraction, cont'd



Interplanar spacing *d* is related to the unit cell dimension a_0

$$5d = \sqrt{\frac{5}{4}a_0^2}$$
 or $d = \frac{a_0}{20} = 0.2236a_0$

Not only can crystals be used to separate different x-ray wavelengths, but x-rays in turn can be used to study crystals, for example determine the type of crystal ordering and $a_{0.}$

Crystal structure, lattice planes, and Miller indices

Planes with different Miller indices in cubic crystals. The *inverse* of these fractional intercepts yields the Miller indices *h*, *k*, *l*.



from http://en.wikipedia.org/wiki/Miller_index

Crystal structure and Miller indices



from http://en.wikipedia.org/wiki/Miller_index

Crystal structure and Miller indices



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Indexing lattice planes

http://www.msm.cam.ac.uk/doitpoms/tlplib/miller_indices/lattice_index.php

Rock salt (cubic) crystal structure



$$d_{hkl} = \frac{a_0}{\sqrt{h^2 + k^2 + l^2}}$$

Structure factor for NaCl:

$$F = \left[f_{Na} + f_{Cl} e^{i\pi(h+k+l)} \right] 1 + e^{i\pi(h+k)} + e^{i\pi(h+l)} + e^{i\pi(k+l)} \right]$$

$$F = 4(f_{Na} + f_{Cl}) \quad \text{if } h, k, l \text{ are even}$$

$$F = 4(f_{Na} - f_{Cl}) \quad \text{if } h, k, l \text{ are odd}$$

$$F = 0 \quad \text{if } h, k, l \text{ are mixed}$$

X-Ray diffraction: a practical approach, by C. Suryanarayana, M. Grant Norton

X-Ray diffraction (XRD) pattern (diffractogram) from NaCl



http://web.pdx.edu/~pmoeck/phy381/Topic5a-XRD.pdf

The value of d, the distance between adjacent planes in the set (hkl), may be found from the following equations.

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Cubic:

$$\frac{1}{d^2} = \frac{h^2 + k^2 + l^2}{a^2}$$
Tetragonal:

$$\frac{1}{d^2} = \frac{h^2 + k^2}{a^2} + \frac{l^2}{c^2}$$
Hexagonal:

$$\frac{1}{d^2} = \frac{4}{3} \left(\frac{h^2 + hk + k^2}{a^2} \right) + \frac{l^2}{c^2}$$
Rhombohedral:

$$\frac{1}{d^2} = \frac{(h^2 + k^2 + l^2)\sin^2 \alpha + 2(hk + kl + hl)(\cos^2 \alpha - \cos \alpha)}{a^2(1 - 3\cos^2 \alpha + 2\cos^3 \alpha)}$$
Orthorhombic:

$$\frac{1}{d^2} = \frac{h^2}{a^2} + \frac{k^2}{b^2} + \frac{l^2}{c^2}$$
Monoclinic:

$$\frac{1}{d} = \frac{1}{a^2} - \frac{(h^2}{a^2} + \frac{k^2\sin^2 \beta}{b^2} + \frac{l^2}{c^2}$$

$$\frac{1}{d^2} = \frac{1}{\sin^2 \beta} \left(\frac{1}{a^2} + \frac{1}{b^2} + \frac{1}{c^2} - \frac{1}{ac} \right)$$

$$Triclinic: \quad \frac{1}{d^2} = \frac{1}{V^2} \left(S_{11}h^2 + S_{22}k^2 + S_{33}l^2 + 2S_{12}hk + 2S_{23}kl + 2S_{13}hl \right)$$

In the equation for triclinic crystals,

V = volume of unit cell $S_{11} = b^2 c^2 \sin^2 \alpha,$ $S_{22} = a^2 c^2 \sin^2 \beta,$ $S_{33} = a^2 b^2 \sin^2 \gamma,$ $S_{12} = abc^2 (\cos \alpha \cos \beta - \cos \gamma),$ $S_{23} = a^2 bc (\cos \beta \cos \gamma - \cos \alpha),$ $S_{13} = ab^2 c (\cos \gamma \cos \alpha - \cos \beta).$

d spacings for tetragonal, hexagonal, orthorhombic crystals

Plane spacings for: Tetragonal: $\frac{1}{d^2} = \frac{h^2 + k^2}{a^2} + \frac{l^2}{a^2}$ (4) Bragg's Law (1): $d = \frac{\lambda}{2\sin\theta_{c}} \quad (1)$ $\frac{1}{d^2} = \frac{4}{3} \left(\frac{h^2 + hk + k^2}{a^2} \right) + \frac{l^2}{a^2} (5)$ Hexagonal: $d_{hkl} = \frac{a}{\sqrt{h^2 + k^2 + l^2}}$ (2) Plane spacing for cubic crystals $\frac{1}{l^2} = \frac{h^2}{r^2} + \frac{k^2}{h^2} + \frac{l^2}{r^2}$ (6) Orthorhombic: Combined (1) and (2): $\left(\frac{n\lambda}{2a}\right)^2 = \frac{\sin^2\theta}{h^2 + k^2 + l^2} \underbrace{\operatorname{or}}_{(3)} \sin^2\theta = \frac{\lambda^2}{4a^2} (h^2 + k^2 + l^2) \begin{bmatrix} 1 \\ 3 \end{bmatrix}$ If crystal is <u>tetragonal</u> with $a=a\neq c$ then (1) and (4) become: $\sin^2\theta = \frac{\lambda^2}{4\pi^2} \left(h^2 + k^2\right) + \frac{\lambda^2}{4\pi^2} \left(l^2\right)$ (7)For a particular incident x-ray wavelength and cubic If crystal is <u>hexagonal</u> with $a=a\neq c$ then (1) and (5) become: crystal of unit cell size a, this equation predicts all possible $\sin^2 \theta = \frac{\lambda^2}{2\sigma^2} (h^2 + k^2 + hk) + \frac{\lambda^2}{4\sigma^2} (l^2)$ Bragg angles at which diffraction can occur from planes (8) (hkl). If crystal is <u>orthorhombic</u> with $a\neq b\neq c$ then (1) and (6): Diffraction planes are determined solely by the shape $\sin^2 \boldsymbol{\theta} = \frac{\boldsymbol{\lambda}^2}{\boldsymbol{\lambda}^2} (\boldsymbol{h}^2) + \frac{\boldsymbol{\lambda}^2}{\boldsymbol{\lambda}^2} (\boldsymbol{k}^2) + \frac{\boldsymbol{\lambda}^2}{\boldsymbol{\lambda}^2} (\boldsymbol{l}^2)$ (9) and size (lattice parameters) of the unit cell.

CdTe crystal structure (zincblende)



http://en.wikipedia.org/wiki/File:Sphalerite-unit-cell-depth-fade-3D-balls.png

CdTe XRD pattern



X-ray diffactograms of thin films at annealing temperatures of a) 350 C, b) 400 C and c) 450 C.

http://www.chalcogen.infim.ro/159_Deivanayaki.pdf

CdTe XRD pattern (intensity vs. 2θ)



CdS XRD pattern (intensity vs. 2θ)



Scherrer Equation (relationship to Shape Factor)

$$\tau = \frac{K\lambda}{\beta\cos\theta}$$

K is the shape factor, λ represents the x-ray wavelength used for the measurement, β is the line width (FWHM) in radians, θ is the Bragg angle (note, this is not the 2 θ angle, just θ), and τ is the mean size of the crystalline domains. The formula yields a lower bound on the possible particle size.

The shape factor enables one to determine the average size of crystal grains within a polycrystalline thin film. Assuming a Gaussian function to fit the peak, the shape factor is 0.9, so that

$$\tau = \frac{0.9\lambda}{\beta\cos\theta}$$

http://en.wikipedia.org/wiki/Scherrer Equation, http://www.eng.uc.edu/~gbeaucag/Classes/XRD/Chapter3html/Chapter3.html

A CdTe XRD Study



Motivation

Determine why some samples appear shiny, and some appear matte.









Calculated Lattice Parameter



H. R. Moutinho, et. al., Proc. 26th IEEE Photovoltaic Specialist Conf., 431-434 (1997)

Lattice Parameter



Orientation factor calculated using:

$I \Sigma I'$	Crain size calculated using	Kλ
$p = \frac{1}{\Sigma I} \times \frac{1}{I'}$	Grain size calculated using:	$\tau = \frac{1}{\beta \cos \theta}$

	Shiny First Location	Shiny Second Location	Matte First Location	Matte Second Location
Orientation Factor <i>p</i> for (111) Orientation	1.32	1.60	2.00	1.90
Average Grain size τ (nanometers)	264.86	302.70	302.70	325.99
Lattice Parameter <i>a</i> (angstroms)	6.4893 ± 0.0010	6.4906 ± 0.0009	6.4903 ± 0.0011	6.4901 ± 0.0009

S. Speakman, *Estimating Crystal Size using XRD.*, http://prism.mit.edu/xray

G. B. Harris, Phil. Mag., 43, 113-123 (1951)

UT Sample Data (Lab #7, 2013)







